# Twinning in chalcostibite revisited <br> Massimo Nespolo 

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# Twinning in chalcostibite revisited 

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#### Abstract

Twinning in chalcostibite has been reported to occur on the (104) plane with respect to the standard setting Pnma of the space group. However, different axial settings have been used for this mineral, and in the literature an axial ratio which corresponds to the Pnam setting is often reported. Unfortunately, the same indexing of the twin plane, (104), is incorrectly used for both settings; this has led to an incorrect reticular analysis of the twin, which could have been an example of diperiodic twin, an alleged category which has no known examples to date. In this article, we trace back the origin of the inconsistent indexing, we present a revised reticular analysis which puts back this twin in the category of triperiodic twins, and we discuss the atomic structure at the composition plane, showing that a small adjustment of the metal-sulphur coordination is required to accommodate the change in orientation occurring when the twin forms.


Key-words: chalcostibite; crystallographic orbits; twinning.

## 1 Introduction

In our review of hybrid twins in non-silicate minerals (Nespolo and Ferraris, 2009) we have shown that almost all non-Friedelian twins, i.e. twins with a twin index beyond the empirical limit of 6, are hybrid twins whose effective twin index is within or close to this limit. One remarkable exception remained however, which seemed to escape the reticular theory of twinning (Friedel, 1926), even in its last extension as theory of hybrid twins (Nespolo \& Ferraris, 2005): (104) twin in chalcostibite $\mathrm{CuSbS}_{2}$. In our analysis, we found two concurrent sublattices with twin indices 29 and 25, leading to an effective twin index of 14.5, far too large to explain this twin on the basis of the reticular
theory . Chalcostibite (104) could therefore be the first example of diperiodic twin, an alleged category introduced as intermediate between the usual triperiodic twins and the exceptional case of monoperiodic twins discovered in Zinnwald twin of quartz (Friedel, 1933).

While working on the relation between diperiodic twins, of which to the best of our knowledge there are no confirmed representatives to date, and plesiotwins, a type of oriented crystal association based on a large coincidence site lattice (Nespolo et al., 1999), we have traced back the history of that (104) twin in chalcostibite and realized that, due to a variety of axial settings adopted by different authors, at a certain point an error in the indexing occurred in the literature, which remains in reference textbooks. In this article, we give a brief historical perspective of this twin, show that once the twin plane is correctly indexed the twin is still hybrid but with a much lower effective twin index ( 5.5 instead of 14.5), and explain its occurrence on the basis of pseudo-eigensymmetry of the crystallographic orbits occupied by Cu and Sb , which is the basis of a satisfactory structural restoration across the twin plane.

## 2 Axial setting of chalcostibite and the problem of indexing the twin and composition plane

Chalcostibite (named by Glocker, 1847, p. 32) is a sulfosalt also reported under different names: kupferantimonglanz (Zincken, 1835); rosite (Huot, 1841, p. 197); wolfsbergite, from the type locality (Wolfsberg, Germany: Nicol, 1849, p. 484); guejarite, from Guejar in Spain (Cumenge, 1879).

Chalcostibite crystallizes in a space group of type Pnma (No. 62); both the point group ( mmm ) and the type of unit cell $(o P)$ are invariant under any permutation of the basis vectors. Such a permutation does result in different settings of the space group (a recent detailed treatment is available in Nespolo and Aroyo, 2016), but the structural information was not available in the old mineralogical literature. Before the discovery of X-ray diffraction, the unit cell was estimated from

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an axial ratio obtained by indexing the developed faces, and expressed in the form of a ratio $a: b: c$, usually normalized with respect to $\boldsymbol{b}$ taken as the longest parameter; however, different orientations resulted in different indexing of the faces. In case of orthorhombic crystals, the presence of symmetry operations of the same order (binary rotations or mirror reflections) along the three basis vectors made the indexing particularly disputable. As a result, different axial ratios were often adopted by various authors. In the case of chalcostibite, the following ratios were reported:

- Friedel (1879): 0.7841:1:0.8220
- Groth (1889): 0.5231:1:0.6395
- Laspeyres (1891): 0.527598:1:0.618584
- Penfield and Frenzel (1897, 1898): 0.5312:1:0.63955 (from their own measurements); 0.5283:1:0.6364 (from measurement of Laspeyres, 1891); 0.5242:1:0.6377 (from measurement of Friedel, 1879, and re-indexing forms $\{730\}$ and $\{370\}$ as $\{320\}$ and $\{230\}$ respectively)
- Goldschmidt (1923): 0.8026:1:0.6275
- Ungemach (1923): 0.6275:1:2.4082
- Palache et al. (1952): 0.4153:1:0.2606

The ratios reported by Groth, by Laspeyers, and by Penfield and Frenzel (GLP in the following, for short) were essentially the same. The transformations are the following ${ }^{1}$ : Friedel to GLP: $\mathbf{c}, 2 \mathbf{a}, \mathbf{b}$; Friedel to Palache: $3 \mathbf{c}, 6 \mathbf{b},-2 \mathbf{a}$; GLP to Palache: $3 \mathbf{a}, 6 \mathbf{c}, \mathbf{- b}$; Goldschmidt to Palache: $\mathbf{b}, 3 \mathbf{a},-\mathbf{c}$; Ungemach to Palache: $\mathbf{b}, \mathbf{c}, \mathbf{a}$.

Cell parameters obtained from X-ray diffraction were published by Hofmann (1933), for the standard setting Pnma of the space group, namely $a=6.12, b=3.89, c=14.51 \AA$, corresponding to the ratio $1.5899: 1: 3.8254$, i.e. $0.4156: 0.2614: 1$, very close to Palache apart from the exchange of $\mathbf{b}$

1 Some of the transformations given by Palache et al. (1952) correspond to a change to left-handed reference. In these cases, we have modified a sign to obtain a right-handed setting.

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and $\mathbf{c}$ basis vectors. Cell parameters from X-ray diffraction were also reported by Cervelle et al. (1979) as $a=6.026, b=14.510, c=3.800 \AA$, without any indication of the space group; they however clearly correspond to the Pnam setting and a ratio $0.4153: 1: 0.2619$, very close to the ratio of Palache. The transformation from Hofmann to Palache is a,c,-b and from Palache to Hofmann is $\mathbf{a , - c}, \mathbf{b}$. Finally, the transformation from GLP to Hoffman is $3 \mathbf{a}, \mathbf{b}, 6 \mathbf{c}$. In other words, the orientation of the basis vectors chosen in GLP is consistent with the cell parameters for the standard setting of the space group, although the ratios differ.

Twinning in chalcostibite was reported by Spencer (1896) based on axial ratio of Groth (1889), as occurring on the (102) plane. The same twin was also listed by Hintze (1904, page 992), who used Penfield's axial ratio. The twin plane is reported to coincide with the composition plane and, with respect to the cell parameters obtained from the X-ray diffraction data, it becomes (104) in the Pnma setting or (140) in the Pnam setting, after removing common factors remaining by adopting the transformation. Palache et al. (1952) adopted basis vectors which correspond to the Pnam setting of the space group; accordingly, the twin plane should have been indicated as (140). Unfortunately, the Miller indices of the twin plane were taken directly from Hofmann (1933); in other words, Palache et al. (1952) gave the axial ratio corresponding to the Pnam setting of the space group, but the Miller indices corresponding to the Pnma setting. This error was replicated in a few texts and databases where the twin is mentioned, for example mindat.org (entry No. 983 for chalcostibite).

In our survey (Nespolo and Ferraris, 2009), based on the data reported by Palache et al. (1952), assuming that the indexing was consistent with the axial ratio, we exchanged the $k$ and $l$ indices when computing the twin lattice. The very large effective twin index was the result of the incorrect indexing of the twin plane in Palache et al. (1952). After discovering this error, we repeated the calculations and we report the correct results here.

## 3 Reticular analysis of the (104) twin in chalcostibite

As we have shown in the previous section, the twin and composition plane reported by Hintze (1904) has Miller indices (140) in the Pnam setting reported by Palache et al., and (104) in the standard Pnma setting. On the basis of the cell parameters reported by Razmara et al. (1997) for Pnma, namely $a=6.0160(1), b=3.7968(6), c=14.499(3) \AA$, two concurrent sublattices are based on the twin plane, with quasi-perpendicular directions [302] (index 11, obliquity $0.83^{\circ}$ ) and [403] (index 8 , obliquity $2.12^{\circ}$ ), which lead to an effective twin index $n_{E}=5.5$, as shown in Fig. 1 (calculations performed with the software geminography: Nespolo and Ferraris, 2006). This result shows that (104) twin in chalcostibite is a normal, triperiodic hybrid twin; the large effective twin index reported in our 2009 survey was the result of the error in the indexing of the twin plane in Palache et al. (1952). The possibility of a diperiodic twin is therefore negated.

## 4 Structural analysis of the (104) twin in chalcostibite

Fig. 2 shows the structure of chalcostibite along the $\mathbf{b}$ axis (black: Cu ; grey: Sb ; white: S ) in the cation-centred description. Cu has a slightly distorted tetrahedral coordination with $\mathrm{Cu}-\mathrm{S}$ bond lengths $2.2717,2 \times 2.3157$ and $2.3163 \AA$, whereas Sb has a more irregular coordination, in the shape of a highly distorted tetragonal pyramid, with five bond distances at $2.4489,2 \times 2.5895$ and $2 \times 3.1288 \AA$; the two longer distances barely contribute to the coordination of Sb . Indeed, the Effective COoordination Number (ECoN: Hoppe, 1979) for Cu and Sb is 3.99 for four $\mathrm{Cu}-\mathrm{S}$ bonds and 3.36 for five $\mathrm{Sb}-\mathrm{S}$ bonds, showing that the coordination for Sb is much more distorted than for Cu and is more realistically described a $3+2$. In the anion-centred description (Fig. 3), one $\mathrm{S}(\mathrm{S} 1)$ is coordinated by two Cu , both at $2.3157 \AA$, and three Sb , at 2.4489 and $2 \times 3.1288 \AA$; again, the two longer distances $\mathrm{S}-\mathrm{Sb}$ are borderline bond contacts. The other $\mathrm{S}(\mathrm{S} 2)$ is coordinated by two Cu at 2.2717 and $2.3163 \AA$, and two Sb , both at $2 \times 2.5895 \AA$ (computations performed with the software

CHARDI2015: Nespolo and Guillot, 2016). The structure of minerals is commonly described as based on cation-centred polyhedra, but the importance of the alternative interpretation as anioncentred polyhedra has been repeatedly emphasized (see, e.g., Krivovichev et al., 2013; Beck, 2014; Eon and Nespolo, 2015). As it was already pointed out by Hofmann (1933), both types of metal atoms are very close to the (104) plane (Fig. 4); their position should therefore not be largely affected by twinning. These atoms are located at at the corners of the coordination polyhedra in the anion-centred description of the structure, which makes the visual comparison of the changes in coordination due to twinning easier. The anion-centred description is therefore privileged below; the cation-centred description leads to the same conclusions and is briefly presented in the discussion.

Hofmann (1933) did not provide any quantitative estimation of the perturbation the structure undergoes at the formation of the twin. To obtain such a quantitative measure, we need to analyse the pseudo-eigensymmetry of the crystallographic orbits, to find whether it contains an operation whose linear part coincides with the twin operation. The approach has been described in details in our previous articles, where the reader can find all the details; here we give just a brief summary. The same approach has allowed us to explain the formation of twins in a number of minerals, namely melilite (Marzouki et al. 2014a), staurolite (Marzouki et al. 2014b), aragonite (Marzouki et al., 2015), marcasite (Nespolo and Souvignier, 2015a), cassiterite and rutile (Nespolo and Souvignier, 2015b), forsterite (Azevedo and Nespolo, 2017) and the Carlsbad twin in orthoclase (Nespolo and Souvignier, 2017).

The unit cell of the twin lattice is defined by the (104) plane and the [302] direction quasiperpendicular to it. Because the twin is hybrid, more than one direction quasi-perpendicular to the (104) plane could be chosen, each corresponding to a different sublattice: in this case, [302] (obliquity $0.83^{\circ}$ ) and [403] (obliquity $2.12^{\circ}$ ). The former direction is chosen to describe the structure of chalcostibite in the setting of the twin, because of the lower obliquity, which allows a

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more precise estimation of the pseudo-symmetry.
The two shortest directions in the (104) plane are [40 $\overline{1}$ ] and [010], which, together with [302], constitute the columns of the transformation matrix from the basis vectors (abc) of the untwinned chalcostibite to those of the $\operatorname{twin}(\mathbf{a b c})_{\mathrm{T}}(\mathrm{I}=$ individual, $\mathrm{T}=\mathrm{twin})$ :

$$
\left(\begin{array}{lll}
\mathbf{a} & \mathbf{b} & \mathbf{c}
\end{array}\right)_{\mathrm{I}}\left(\begin{array}{ccc}
4 & 0 & 3 \\
0 & 1 & 0 \\
\frac{1}{1} & 0 & 2
\end{array}\right)=\left(\begin{array}{lll}
\mathbf{a} & \mathbf{b} & \mathbf{c}
\end{array}\right)_{\mathrm{T}}
$$

i.e. $\mathbf{a}_{\mathrm{T}} \mathbf{b}_{\mathrm{T}} \mathbf{c}_{\mathrm{T}}=4 \mathbf{a}_{\mathrm{I}}-\mathbf{c}_{\mathbf{c}}, \mathbf{b}_{\mathrm{I}}, 3 \mathbf{a}_{1}+2 \mathbf{c}_{\mathrm{l}}$. The cell parameters of the twin lattice are $a_{\mathrm{T}}=28.094, b_{\mathrm{T}}=3.797, c_{\mathrm{T}}$ $=34.156 \AA, \beta_{\mathrm{T}}=89.17^{\circ}$; the unit cell of the twin lattice is primitive, monoclinic but pseudoorthorhombic, with a deviation of $\beta$ which corresponds precisely to the obliquity and is eleven times larger than the unit cell of the individual (this correspond to the index of the first sublattice); it contains $16 \times 11=176$ atoms. The highest subgroup of Pnma compatible with this lattice is obtained as $\mathcal{H}=G_{1} \cap G_{2}$ where $G_{1}$ and $G_{2}$ are the space groups of the two individuals, both of type Pnma, in the respective orientations. In our case, $\mathcal{H}$ is of type $P 2_{1} / m$. The fractional coordinates of chalcostibite in the asymmetric unit of Pnma are given in Table 1. The Wyckoff splitting scheme to the setting of the twin lattice is very simple: $4 c \rightarrow 22 \times 2 e$. Table 2 [Note that Table 2 is deposited, not printed, and freely available as Supplementary Material linked to this article on the GSW website of the journal: http://eurjmin.geoscienceworld.org/] gives the 88 atoms in the asymmetric unit of $P 2_{1} / m$. Crystallographic orbits having a (pseudo)-eigensymmetry containing a mirror or glide plane perpendicular to the $\mathbf{c}$ vector of the twin lattice are (quasi)-restored by the corresponding operation. In the cell of the twin lattice, there are 11 positions for the (001) plane passing on or close to the metal atoms, about $3.1 \AA$ apart. The eigensymmetry analysis can be performed about any of these positions: we take the one about $z=0.5$. The position of the metal atoms about $z=0.5$ are pseudosymmetric with respect to two orthorhombic minimal (index 2) supergroups of $P 2_{1} / m$ :

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Pmmn (No. 59), with the basis vector in common with the individual, and Pbmm, which corresponds to Pmma (No. 51) expressed in a non-standard setting obtained with basis vector transformation $\mathbf{c}, \mathbf{a}, \mathbf{b}$. Table 3 gives the fractional atomic coordinates of the four independent metal atoms (the other four are obtained by applying an inversion), as well as the corresponding fractional atomic coordinates obtained by applying an $n$-glide reflection (for Pmmn pseudo-eigensymmetry) and a mirror reflection (for Pbmm pseudo-eigensymmetry), both about $z=0.5$. The Sb atoms are restored by this operation within, respectively, $0.2625 \AA$ and $0.2732 \AA$, the Cu atoms within 0.3849 $\AA$ and $0.3825 \AA$. This means that if a change of orientation occurs during the crystal growth at $z=$ 0.5 (with respect to the unit cell of the twin lattice), only a small adjustment of the metal atoms is necessary to overcome this perturbation. The two candidate restoration operations corresponding to the two pseudo-eigensymmetries do not lead, however, to the same coordination environment on the two sides of the twin plane. The situation is shown in Fig. 5, where the region around $z=0.5$ is shown for the untwinned crystal (bottom), the twinned model in Pmmn (middle), and the twinned model in Pbmm (top). The position of the metal atoms on the twin plan being slightly different in the two individuals, they are shown as split, almost overlapped atoms in the twinned models in Fig. 5. The coordination polyhedra share edges on the twin plane in the Pmmn model, as in the untwinned crystal, whereas they share faces in the Pbmm model. Twinning by reflection on (104) requires therefore a small adjustment of the metal atoms to produce a local structure at the boundary which is close to that of the untwinned crystal if the restoration operation is an $n$-glide; if instead the restoration operation is a mirror reflection, the coordination that results highly modified, an event which would make the twin less likely to occur.

Now, exactly the same conclusion can be drawn for the eleven positions of the twin plan every $\sim 3.1 \AA$. This means that during the crystal growth, twinning on (104) can be accommodated every $\sim 3.1 \AA$, making this twinning a relatively high probability event.

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## 5 Discussion

Fig 6 is obtained from the Pmmn twin model by taking the arithmetic average of the fractional coordinates of the metal atoms almost overlapped in the Fig. 5 (middle). The coordination polyhedra are essentially the same as those occurring in the untwinned mineral, but the S-metal distances are in some cases severely affected, resulting in some polyhedra of unphysically small size, with some S-metal distances as short at $1.6 \AA$. Clearly, the structural adjustment has to extend further from the composition plane in order for the S-metal distances to approach their normal values. From the energetic viewpoint, it is certainly more favourable if the strain occurring at the composition plane is absorbed through smaller deformations on a slightly wider region than a single larger deformation at the composition plane.

Fig. 7 is the cation-centred description of the structure of chalcostibite analogous to Fig. 5. The coordination polyhedra at the twin plane for the Pmmn model are essentially the same as those in the untwinned crystal, with the exception of one Cu site; here, one S has moved closer and another one farther from Cu , leading to a triangular instead of tetrahedral coordination. The structural adjustment close to the twin plane mentioned above would easily restore the normal coordination of this Cu atom. In the Pbmm model, instead, the coordination at the twin plane is so heavily affected that it would require extensive rearrangements to get closer to the normal coordination in the bulk of the crystal. The Pmmn model is definitely more sound than the Pbmm model and the restoration operation for this twin is an $n$-glide perpendicular to the c axis of the twin lattice.

The determination of the extent and extension of the structural adjustment around the twin plane would require a high-resolution transmission electron microscopy study of (104) twin in chalcostibite, which, to the best of our knowledge, has never been performed. With this study, we hope to draw the attention of electron microscopists on the interest to perform such an investigation.

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## FIGURE CAPTIONS

Figure 1. The lattice of chalcostibite projected along the [010] direction. drawn in scale according to the cell parameters reported by Razmara et al. (1997). Four-pointed and five-pointed black stars indicate lattice nodes belonging to the [302]/(104) and [403]/(104) sublattice respectively; nodes on the (104) plane are common the two two sublattice and indicated by black circle; white circles stand for lattice nodes not belonging to either of the sublattices. The unit cell based on the [302]/(104) pair of lattice elements has multiplicity 11 and contains one lattice node for each sublattice, so that the effective twin index is $11 / 2=5.5$.
Figure 2. The structure of chalcostibite observed down the $\mathbf{b}$ axis in the cation-centred description. Cu atoms occupy a distorted tetrahedron whereas Sb atoms are at the centre of a face of a distorted square pyramid (this and the following figures are drawn with VESTA: Momma and Izumi, 2011).
Figure 3. The structure of chalcostibite observed down the b axis in the anion-centred description. The two crystallographically independent $S$ atoms have different coordination: a $2 \mathrm{Cu}+3 \mathrm{Sb}$ distorted square pyramid and a $2 \mathrm{Cu}+2 \mathrm{Sb}$ distorted tetrahedron.
Figure 4. The ball-and-stick description of the structure of chalcostibite, with the (104) plane drawn in grey at the position $z=0.5$. Metal atoms are positioned very close to this plane. Cu atoms are shown in black, Sb atoms in grey, S atoms in white.
Figure 5. A portion of the structure of chalcostibite in the axial setting of the twin, drawn around the twin plane in the anion-centred description. Bottom: the untwinned structure. Middle: the Pmmn twinned model (the portion above the twin plane is obtained from the one below it by an $n$-glide reflection at $z=0.5$ ). Top: the $P b m m$ twinned model (the portion above the twin plane is obtained from the one below it by a mirror reflection at $z=0.5$ ). In both twinned models, the metal atoms close to the twin plane are shown in the positions corresponding to the two individuals; these positions being very close, the corresponding atoms are almost overlapped in the figure. Whereas in the Pnnm model the coordination polyhedra on the opposite sides of the twin plane share their edges, as in the untwinned crystal, in the Pbmm model they share their faces, leading to a physically unlikely situation.
Fig. 6. A likely structural adjustment at the composition plane, obtained by taking the arithmetic average of the fractional atomic coordinates of the metal atoms quasi-overlapped in Fig. 5. The Smetal coordination appears very similar to that in the untwinned crystal but actually some of the polyhedra have unphysically small volumes, indicating that the structural adjustment likely extends for a few atomic planes in order to minimize the structural strain occurring when the twin forms.
Figure 7. A portion of the structure of chalcostibite in the axial setting of the twin, drawn around the twin plane in the cation-centred description. Bottom: the untwinned structure; middle: the Pmmn twinned model; top: the Pbmm twinned model (same conventions as in Fig. 5). The positions of the Cu and Sb atoms on the twin plane are the arithmetic average of the positions in the two individuals; no other structural adjustment is introduced. In the Pnnm model, the coordination polyhedra are essentially the same as in the untwinned crystal, with the exception of one of the Cu sites; in the Pbmm model, instead, the coordination around the twin plane is severely affected. A structural adjustment of the S atoms close to the twin plane can restore the normal coordination in the Pmmn model but not in the Pbmm model.

Table 1. Fractional atomic coordinates in the asymmetric unit of chalcostibite, setting Pnma. All atoms are in Wyckoff position 4c. Data from Ramzara et al (1997).

| Cu | $0.7463(6)$ | $3 / 4$ | $0.1721(2)$ |
| :--- | ---: | ---: | ---: |
| Sb | $0.2245(2)$ | $1 / 4$ | $0.0636(1)$ |
| S 1 | $0.624(1)$ | $1 / 4$ | $0.0960(4)$ |
| S 2 | $0.131(2)$ | $3 / 4$ | $0.1787(5)$ |

Table 3. Fractional atomic coordinates of metal atoms around $z=0.5$ (label without ') and after application of a mirror reflection about $z=0.5$ label with '), in the two models Pmmn and Pbmm. The difference is expressed both in terms of fractional atomic coordinated and in $\AA(|\Delta|)$.

| Pmmn |  |  |  |  |  |  |  |  |  |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Atom | $x$ | $y$ | z | Atom | $x$ | $y$ | z | $\Delta x$ | $\Delta y$ | $\Delta z$ | $\|\Delta\|$ |
| Sb11 | 0.93256 | 0.25 | 0.49808 | Sb18' | 0.94184 | 0.25 | 0.49728 | 0.00928 | 0 | 0.0008 | 0.2625 |
| Sb18 | 0.44184 | 0.75 | 0.50272 | Sb11' | 0.43256 | 0.75 | 0.50192 | 0.00928 | 0 | 0.0008 | 0.2625 |
| Cu20 | 0.68263 | 0.75 | 0.50526 | Cu20' | 0.68397 | 0.75 | 0.49406 | 0.00134 | 0 | 0.0112 | 0.3849 |
| Cu 2 | 0.18397 | 0.25 | 0.50594 | $\mathrm{Cu} 2{ }^{\prime}$ | 0.18263 | 0.25 | 0.49474 | 0.00134 | 0 | 0.0112 | 0.3849 |
| Pbmm |  |  |  |  |  |  |  |  |  |  |  |
| Atom | $x$ | $y$ | z | Atom | $x$ | $y$ | z | $\Delta x$ | $\Delta y$ | $\Delta z$ | $\|\Delta\|$ |
| Sb11 | 0.93256 | 0.25 | 0.49808 | Sb11' | 0.93256 | 0.25 | 0.50192 | 0 | 0 | 0.0008 | 0.2732 |
| Sb18 | 0.44184 | 0.75 | 0.50272 | Sb18' | 0.44184 | 0.75 | 0.49728 | 0 | 0 | 0.0008 | 0.2732 |
| Cu20 | 0.68263 | 0.75 | 0.50526 | $\mathrm{Cu} 2{ }^{\prime}$ | 0.68263 | 0.75 | 0.49474 | 0 | 0 | 0.0112 | 0.3825 |
| Cu 2 | 0.18397 | 0.25 | 0.50594 | $\mathrm{Cu} 02{ }^{\prime}$ | 0.18397 | 0.25 | 0.49406 | 0 | 0 | 0.0112 | 0.3825 |










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