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Twinning in chalcostibite revisited

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Abstract: Twinning in chalcostibite has been reported to occur on the (104) plane with respect to 6 the standard setting Pnma of the space group. However, different axial settings have been used for 7 this mineral, and in the literature an axial ratio which corresponds to the Pnam setting is often 8 9 reported. Unfortunately, the same indexing of the twin plane, (104), is incorrectly used for both settings; this has led to an incorrect reticular analysis of the twin, which could have been an 10 11 example of diperiodic twin, an alleged category which has no known examples to date. In this 12 article, we trace back the origin of the inconsistent indexing, we present a revised reticular analysis 13 which puts back this twin in the category of triperiodic twins, and we discuss the atomic structure at 14 the composition plane, showing that a small adjustment of the metal-sulphur coordination is 15 required to accommodate the change in orientation occurring when the twin forms.

16 **Key-words:** chalcostibite; crystallographic orbits; twinning.

17 1 Introduction

In our review of hybrid twins in non-silicate minerals (Nespolo and Ferraris, 2009) we have shown that almost all non-Friedelian twins, *i.e.* twins with a twin index beyond the empirical limit of 6, are hybrid twins whose effective twin index is within or close to this limit. One remarkable exception remained however, which seemed to escape the reticular theory of twinning (Friedel, 1926), even in its last extension as theory of hybrid twins (Nespolo & Ferraris, 2005): (104) twin in chalcostibite CuSbS₂. In our analysis, we found two concurrent sublattices with twin indices 29 and 25, leading to an effective twin index of 14.5, far too large to explain this twin on the basis of the reticular

theory . Chalcostibite (104) could therefore be the first example of diperiodic twin, an alleged category introduced as intermediate between the usual triperiodic twins and the exceptional case of monoperiodic twins discovered in Zinnwald twin of quartz (Friedel, 1933).

28 While working on the relation between diperiodic twins, of which to the best of our knowledge 29 there are no confirmed representatives to date, and plesiotwins, a type of oriented crystal association based on a large coincidence site lattice (Nespolo et al., 1999), we have traced back the history of 30 that (104) twin in chalcostibite and realized that, due to a variety of axial settings adopted by 31 32 different authors, at a certain point an error in the indexing occurred in the literature, which remains 33 in reference textbooks. In this article, we give a brief historical perspective of this twin, show that 34 once the twin plane is correctly indexed the twin is still hybrid but with a much lower effective twin 35 index (5.5 instead of 14.5), and explain its occurrence on the basis of pseudo-eigensymmetry of the 36 crystallographic orbits occupied by Cu and Sb, which is the basis of a satisfactory structural 37 restoration across the twin plane.

Axial setting of chalcostibite and the problem of indexing the twin and composition plane

Chalcostibite (named by Glocker, 1847, p. 32) is a sulfosalt also reported under different names: *kupferantimonglanz* (Zincken, 1835); *rosite* (Huot, 1841, p. 197); *wolfsbergite*, from the type
locality (Wolfsberg, Germany: Nicol, 1849, p. 484); *guejarite*, from Guejar in Spain (Cumenge, 1879).

44 Chalcostibite crystallizes in a space group of type *Pnma* (No. 62); both the point group (*mmm*) and 45 the type of unit cell (*oP*) are invariant under any permutation of the basis vectors. Such a 46 permutation does result in different settings of the space group (a recent detailed treatment is 47 available in Nespolo and Aroyo, 2016), but the structural information was not available in the old 48 mineralogical literature. Before the discovery of X-ray diffraction, the unit cell was estimated from

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49 an axial ratio obtained by indexing the developed faces, and expressed in the form of a ratio *a:b:c*, 50 *usually* normalized with respect to *b* taken as the longest parameter; however, different orientations 51 resulted in different indexing of the faces. In case of orthorhombic crystals, the presence of 52 symmetry operations of the same order (binary rotations or mirror reflections) along the three basis 53 vectors made the indexing particularly disputable. As a result, different axial ratios were often 54 adopted by various authors. In the case of chalcostibite, the following ratios were reported:

- Friedel (1879): 0.7841:1:0.8220
- Groth (1889): 0.5231:1:0.6395
- Laspeyres (1891): 0.527598:1:0.618584
- Penfield and Frenzel (1897, 1898): 0.5312:1:0.63955 (from their own measurements);
 0.5283:1:0.6364 (from measurement of Laspeyres, 1891); 0.5242:1:0.6377 (from measurement of Friedel, 1879, and re-indexing forms {730} and {370} as {320} and {230}
 respectively)
- 62 Goldschmidt (1923): 0.8026:1:0.6275
- Ungemach (1923): 0.6275:1:2.4082
- Palache *et al.* (1952): 0.4153:1:0.2606

The ratios reported by Groth, by Laspeyers, and by Penfield and Frenzel (GLP in the following, for short) were essentially the same. The transformations are the following¹: Friedel to GLP: **c**,2**a**,**b**; Friedel to Palache: 3**c**,6**b**,-2**a**; GLP to Palache: 3**a**,6**c**,-**b**; Goldschmidt to Palache: **b**,3**a**,-**c**; Ungemach to Palache: **b**,**c**,**a**.

- 69 Cell parameters obtained from X-ray diffraction were published by Hofmann (1933), for the
- standard setting *Pnma* of the space group, namely a = 6.12, b = 3.89, c = 14.51 Å, corresponding to
- the ratio 1.5899:1:3.8254, *i.e.* 0.4156:0.2614:1, very close to Palache apart from the exchange of **b**
- 1 Some of the transformations given by Palache *et al.* (1952) correspond to a change to left-handed reference.
 In these cases, we have modified a sign to obtain a right-handed setting.
- 7 8

and **c** basis vectors. Cell parameters from X-ray diffraction were also reported by Cervelle *et al.* (1979) as a = 6.026, b = 14.510, c = 3.800 Å, without any indication of the space group; they however clearly correspond to the *Pnam* setting and a ratio 0.4153:1:0.2619, very close to the ratio of Palache. The transformation from Hofmann to Palache is **a**,**c**,-**b** and from Palache to Hofmann is **a**,-**c**,**b**. Finally, the transformation from GLP to Hoffman is 3**a**,**b**,6**c**. In other words, the *orientation* of the basis vectors chosen in GLP is consistent with the cell parameters for the standard setting of the space group, although the ratios differ.

79 Twinning in chalcostibite was reported by Spencer (1896) based on axial ratio of Groth (1889), as 80 occurring on the (102) plane. The same twin was also listed by Hintze (1904, page 992), who used 81 Penfield's axial ratio. The twin plane is reported to coincide with the composition plane and, with respect to the cell parameters obtained from the X-ray diffraction data, it becomes (104) in the 82 83 Pnma setting or (140) in the Pnam setting, after removing common factors remaining by adopting 84 the transformation. Palache et al. (1952) adopted basis vectors which correspond to the Pnam 85 setting of the space group; accordingly, the twin plane should have been indicated as (140). 86 Unfortunately, the Miller indices of the twin plane were taken directly from Hofmann (1933); in 87 other words, Palache et al. (1952) gave the axial ratio corresponding to the Pnam setting of the 88 space group, but the Miller indices corresponding to the Pnma setting. This error was replicated in a 89 few texts and databases where the twin is mentioned, for example mindat.org (entry No. 983 for 90 chalcostibite).

In our survey (Nespolo and Ferraris, 2009), based on the data reported by Palache *et al.* (1952), assuming that the indexing was consistent with the axial ratio, we exchanged the k and l indices when computing the twin lattice. The very large effective twin index was the result of the incorrect indexing of the twin plane in Palache *et al.* (1952). After discovering this error, we repeated the calculations and we report the correct results here.

96 **3** Reticular analysis of the (104) twin in chalcostibite

97 As we have shown in the previous section, the twin and composition plane reported by Hintze 98 (1904) has Miller indices (140) in the Pnam setting reported by Palache et al., and (104) in the 99 standard Pnma setting. On the basis of the cell parameters reported by Razmara et al. (1997) for *Pnma*, namely a = 6.0160(1), b = 3.7968(6), c = 14.499(3) Å, two concurrent sublattices are based 100 101 on the twin plane, with quasi-perpendicular directions [302] (index 11, obliquity 0.83°) and [403] (index 8, obliquity 2.12°), which lead to an effective twin index $n_{\rm E} = 5.5$, as shown in Fig. 1 102 103 (calculations performed with the software geminography: Nespolo and Ferraris, 2006). This result 104 shows that (104) twin in chalcostibite is a normal, triperiodic hybrid twin; the large effective twin 105 index reported in our 2009 survey was the result of the error in the indexing of the twin plane in 106 Palache et al. (1952). The possibility of a diperiodic twin is therefore negated.

107 4 Structural analysis of the (104) twin in chalcostibite

108 Fig. 2 shows the structure of chalcostibite along the b axis (black: Cu; grey: Sb; white: S) in the 109 cation-centred description. Cu has a slightly distorted tetrahedral coordination with Cu-S bond lengths 2.2717, 2×2.3157 and 2.3163 Å, whereas Sb has a more irregular coordination, in the shape 110 111 of a highly distorted tetragonal pyramid, with five bond distances at 2.4489, 2×2.5895 and 2×3.1288 Å; the two longer distances barely contribute to the coordination of Sb. Indeed, the 112 Effective COoordination Number (ECoN: Hoppe, 1979) for Cu and Sb is 3.99 for four Cu-S bonds 113 and 3.36 for five Sb-S bonds, showing that the coordination for Sb is much more distorted than for 114 115 Cu and is more realistically described a 3+2. In the anion-centred description (Fig. 3), one S (S1) is coordinated by two Cu, both at 2.3157 Å, and three Sb, at 2.4489 and 2×3.1288 Å; again, the two 116 longer distances S-Sb are borderline bond contacts. The other S (S2) is coordinated by two Cu at 117 2.2717 and 2.3163 Å, and two Sb, both at 2×2.5895 Å (computations performed with the software 118

119 CHARDI2015: Nespolo and Guillot, 2016). The structure of minerals is commonly described as 120 based on cation-centred polyhedra, but the importance of the alternative interpretation as anion-121 centred polyhedra has been repeatedly emphasized (see, e.g., Krivovichev et al., 2013; Beck, 2014; 122 Eon and Nespolo, 2015). As it was already pointed out by Hofmann (1933), both types of metal 123 atoms are very close to the (104) plane (Fig. 4); their position should therefore not be largely 124 affected by twinning. These atoms are located at at the corners of the coordination polyhedra in the 125 anion-centred description of the structure, which makes the visual comparison of the changes in 126 coordination due to twinning easier. The anion-centred description is therefore privileged below; the 127 cation-centred description leads to the same conclusions and is briefly presented in the discussion.

128 Hofmann (1933) did not provide any quantitative estimation of the perturbation the structure 129 undergoes at the formation of the twin. To obtain such a quantitative measure, we need to analyse 130 the pseudo-eigensymmetry of the crystallographic orbits, to find whether it contains an operation 131 whose linear part coincides with the twin operation. The approach has been described in details in 132 our previous articles, where the reader can find all the details; here we give just a brief summary. 133 The same approach has allowed us to explain the formation of twins in a number of minerals, 134 namely melilite (Marzouki et al. 2014a), staurolite (Marzouki et al. 2014b), aragonite (Marzouki et al., 2015), marcasite (Nespolo and Souvignier, 2015a), cassiterite and rutile (Nespolo and 135 136 Souvignier, 2015b), forsterite (Azevedo and Nespolo, 2017) and the Carlsbad twin in orthoclase 137 (Nespolo and Souvignier, 2017).

The unit cell of the twin lattice is defined by the (104) plane and the [302] direction quasiperpendicular to it. Because the twin is hybrid, more than one direction quasi-perpendicular to the (104) plane could be chosen, each corresponding to a different sublattice: in this case, [302] (obliquity 0.83°) and [403] (obliquity 2.12°). The former direction is chosen to describe the structure of chalcostibite in the setting of the twin, because of the lower obliquity, which allows a

143 more precise estimation of the pseudo-symmetry.

144 The two shortest directions in the (104) plane are $[40\overline{1}]$ and [010], which, together with [302], 145 constitute the columns of the transformation matrix from the basis vectors $(\mathbf{abc})_{I}$ of the untwinned 146 chalcostibite to those of the twin $(\mathbf{abc})_{T}$ (I = individual, T = twin):

$$(\mathbf{a} \quad \mathbf{b} \quad \mathbf{c})_{\mathrm{I}} \begin{pmatrix} 4 & 0 & 3 \\ 0 & 1 & 0 \\ \overline{1} & 0 & 2 \end{pmatrix} = (\mathbf{a} \quad \mathbf{b} \quad \mathbf{c})_{\mathrm{T}}$$

i.e. $\mathbf{a}_{\mathrm{T}}\mathbf{b}_{\mathrm{T}}\mathbf{c}_{\mathrm{T}} = 4\mathbf{a}_{\mathrm{I}}-\mathbf{c}_{\mathrm{I}}, \mathbf{b}_{\mathrm{I}}, 3\mathbf{a}_{\mathrm{I}}+2\mathbf{c}_{\mathrm{I}}$. The cell parameters of the twin lattice are $a_{\mathrm{T}} = 28.094$, $b_{\mathrm{T}} = 3.797$, c_{T} 147 = 34.156 Å, β_T = 89.17°; the unit cell of the twin lattice is primitive, monoclinic but pseudo-148 orthorhombic, with a deviation of β which corresponds precisely to the obliquity and is eleven times 149 150 larger than the unit cell of the individual (this correspond to the index of the first sublattice); it 151 contains $16 \times 11 = 176$ atoms. The highest subgroup of *Pnma* compatible with this lattice is 152 obtained as $\mathcal{H} = \mathcal{G}_1 \cap \mathcal{G}_2$ where \mathcal{G}_1 and \mathcal{G}_2 are the space groups of the two individuals, both of type 153 *Pnma*, in the respective orientations. In our case, \mathcal{H} is of type $P2_1/m$. The fractional coordinates of chalcostibite in the asymmetric unit of Pnma are given in Table 1. The Wyckoff splitting scheme to 154 the setting of the twin lattice is very simple: $4c \rightarrow 22 \times 2e$. Table 2 [Note that Table 2 is deposited, 155 156 not printed, and freely available as Supplementary Material linked to this article on the GSW website of the journal: http://eurjmin.geoscienceworld.org/] gives the 88 atoms in the asymmetric 157 158 unit of $P2_1/m$. Crystallographic orbits having a (pseudo)-eigensymmetry containing a mirror or 159 glide plane perpendicular to the c vector of the twin lattice are (quasi)-restored by the corresponding 160 operation. In the cell of the twin lattice, there are 11 positions for the (001) plane passing on or close to the metal atoms, about 3.1Å apart. The eigensymmetry analysis can be performed about 161 162 any of these positions: we take the one about z = 0.5. The position of the metal atoms about z = 0.5are pseudosymmetric with respect to two orthorhombic minimal (index 2) supergroups of $P2_1/m$: 163

164 Pmmn (No. 59), with the basis vector in common with the individual, and Pbmm, which 165 corresponds to Pmma (No. 51) expressed in a non-standard setting obtained with basis vector 166 transformation c,a,b. Table 3 gives the fractional atomic coordinates of the four independent metal 167 atoms (the other four are obtained by applying an inversion), as well as the corresponding fractional 168 atomic coordinates obtained by applying an *n*-glide reflection (for *Pmmn* pseudo-eigensymmetry) and a mirror reflection (for *Pbmm* pseudo-eigensymmetry), both about z = 0.5. The Sb atoms are 169 restored by this operation within, respectively, 0.2625 Å and 0.2732 Å, the Cu atoms within 0.3849 170 171 Å and 0.3825 Å. This means that if a change of orientation occurs during the crystal growth at z =172 0.5 (with respect to the unit cell of the twin lattice), only a small adjustment of the metal atoms is 173 necessary to overcome this perturbation. The two candidate restoration operations corresponding to 174 the two pseudo-eigensymmetries do not lead, however, to the same coordination environment on the 175 two sides of the twin plane. The situation is shown in Fig. 5, where the region around z = 0.5 is 176 shown for the untwinned crystal (bottom), the twinned model in Pmmn (middle), and the twinned 177 model in *Pbmm* (top). The position of the metal atoms on the twin plan being slightly different in 178 the two individuals, they are shown as split, almost overlapped atoms in the twinned models in Fig. 179 5. The coordination polyhedra share edges on the twin plane in the Pmmn model, as in the 180 untwinned crystal, whereas they share faces in the *Pbmm* model. Twinning by reflection on (104) 181 requires therefore a small adjustment of the metal atoms to produce a local structure at the boundary 182 which is close to that of the untwinned crystal if the restoration operation is an *n*-glide; if instead the 183 restoration operation is a mirror reflection, the coordination that results highly modified, an event 184 which would make the twin less likely to occur.

Now, exactly the same conclusion can be drawn for the eleven positions of the twin plan every
~3.1Å. This means that during the crystal growth, twinning on (104) can be accommodated every
~3.1Å, making this twinning a relatively high probability event.

188 **5 Discussion**

189 Fig 6 is obtained from the *Pmmn* twin model by taking the arithmetic average of the fractional coordinates of the metal atoms almost overlapped in the Fig. 5 (middle). The coordination 190 191 polyhedra are essentially the same as those occurring in the untwinned mineral, but the S-metal 192 distances are in some cases severely affected, resulting in some polyhedra of unphysically small 193 size, with some S-metal distances as short at 1.6 Å. Clearly, the structural adjustment has to extend 194 further from the composition plane in order for the S-metal distances to approach their normal 195 values. From the energetic viewpoint, it is certainly more favourable if the strain occurring at the 196 composition plane is absorbed through smaller deformations on a slightly wider region than a single 197 larger deformation at the composition plane.

198 Fig. 7 is the cation-centred description of the structure of chalcostibite analogous to Fig. 5. The 199 coordination polyhedra at the twin plane for the *Pmmn* model are essentially the same as those in 200 the untwinned crystal, with the exception of one Cu site; here, one S has moved closer and another 201 one farther from Cu, leading to a triangular instead of tetrahedral coordination. The structural 202 adjustment close to the twin plane mentioned above would easily restore the normal coordination of 203 this Cu atom. In the Pbmm model, instead, the coordination at the twin plane is so heavily affected 204 that it would require extensive rearrangements to get closer to the normal coordination in the bulk 205 of the crystal. The Pmmn model is definitely more sound than the Pbmm model and the restoration 206 operation for this twin is an *n*-glide perpendicular to the c axis of the twin lattice.

The determination of the extent and extension of the structural adjustment around the twin plane would require a high-resolution transmission electron microscopy study of (104) twin in chalcostibite, which, to the best of our knowledge, has never been performed. With this study, we hope to draw the attention of electron microscopists on the interest to perform such an investigation. Acknowlegments. We thank Prof. Dominik Schaniel (CRM2, Université de Lorraine) for help with the German literature. Critical remarks from two anonymous reviewers and the Associate Editor are thankfully acknowledged.

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284 FIGURE CAPTIONS

- **Figure 1**. The lattice of chalcostibite projected along the [010] direction. drawn in scale according to the cell parameters reported by Razmara *et al.* (1997). Four-pointed and five-pointed black stars indicate lattice nodes belonging to the [302]/(104) and [403]/(104) sublattice respectively; nodes on the (104) plane are common the two two sublattice and indicated by black circle; white circles stand for lattice nodes not belonging to either of the sublattices. The unit cell based on the [302]/(104)
- pair of lattice elements has multiplicity 11 and contains one lattice node for each sublattice, so that the effective twin index is 11/2 = 5.5.
- Figure 2. The structure of chalcostibite observed down the **b** axis in the cation-centred description. Cu atoms occupy a distorted tetrahedron whereas Sb atoms are at the centre of a face of a distorted square pyramid (this and the following figures are drawn with VESTA: Momma and Izumi, 2011).
- Figure 3. The structure of chalcostibite observed down the b axis in the anion-centred description.
- The two crystallographically independent S atoms have different coordination: a 2Cu+3Sb distorted square pyramid and a 2Cu+2Sb distorted tetrahedron.
- Figure 4. The ball-and-stick description of the structure of chalcostibite, with the (104) plane drawn in grey at the position z = 0.5. Metal atoms are positioned very close to this plane. Cu atoms are shown in black, Sb atoms in grey, S atoms in white.
- shown in black, Sb atoms in grey, S atoms in white. **Figure 5.** A portion of the structure of chalcostibite in the axial setting of the twin, drawn around the twin plane in the anion-centred description. Bottom: the untwinned structure. Middle: the *Pmmn* twinned model (the portion above the twin plane is obtained from the one below it by an *n*-glide reflection at n = 0.5). Then the *Plane* twinned model (the portion shows the twin plane is obtained
- reflection at z = 0.5). Top: the *Pbmm* twinned model (the portion above the twin plane is obtained from the one below it by a mirror reflection at z = 0.5). In both twinned models, the metal atoms close to the twin plane are shown in the positions corresponding to the two individuals; these positions being very close, the corresponding atoms are almost overlapped in the figure. Whereas in the *Pnnm* model the coordination polyhedra on the opposite sides of the twin plane share their edges, as in the untwinned crystal, in the *Pbmm* model they share their faces, leading to a physically unlikely situation.
- **Fig. 6.** A likely structural adjustment at the composition plane, obtained by taking the arithmetic average of the fractional atomic coordinates of the metal atoms quasi-overlapped in Fig. 5. The Smetal coordination appears very similar to that in the untwinned crystal but actually some of the polyhedra have unphysically small volumes, indicating that the structural adjustment likely extends for a few atomic planes in order to minimize the structural strain occurring when the twin forms.
- Figure 7. A portion of the structure of chalcostibite in the axial setting of the twin, drawn around 316 the twin plane in the cation-centred description. Bottom: the untwinned structure; middle: the 317 Pmmn twinned model; top: the Pbmm twinned model (same conventions as in Fig. 5). The 318 positions of the Cu and Sb atoms on the twin plane are the arithmetic average of the positions in the 319 two individuals; no other structural adjustment is introduced. In the Pnnm model, the coordination 320 polyhedra are essentially the same as in the untwinned crystal, with the exception of one of the Cu 321 322 sites; in the Pbmm model, instead, the coordination around the twin plane is severely affected. A 323 structural adjustment of the S atoms close to the twin plane can restore the normal coordination in the *Pmmn* model but not in the *Pbmm* model. 324

Table 1. Fractional atomic coordinates in the asymmetric unit of chalcostibite, setting *Pnma*. All atoms are in Wyckoff position 4*c*. Data from Ramzara *et al* (1997).

Cu0.7463(6)3/40.1721(2)Sb0.2245(2)1/40.0636(1)S10.624(1)1/40.0960(4)S20.131(2)3/40.1787(5)

327 **Table 3.** Fractional atomic coordinates of metal atoms around z = 0.5 (label without ') and after

- 328 application of a mirror reflection about z = 0.5 label with '), in the two models *Pmmn* and *Pbmm*.
- 329 The difference is expressed both in terms of fractional atomic coordinated and in Å ($|\Delta|$).

Pmmn													
Atom	x	у	Z.	Atom	x	у	Z.	Δx	Δy	Δz	$ \Delta $		
Sb11	0.93256	0.25	0.49808	Sb18'	0.94184	0.25	0.49728	0.00928	0	0.0008	0.2625		
Sb18	0.44184	0.75	0.50272	Sb11'	0.43256	0.75	0.50192	0.00928	0	0.0008	0.2625		
Cu20	0.68263	0.75	0.50526	Cu20'	0.68397	0.75	0.49406	0.00134	0	0.0112	0.3849		
Cu2	0.18397	0.25	0.50594	Cu2'	0.18263	0.25	0.49474	0.00134	0	0.0112	0.3849		
Pbmm													
Atom	x	у	Ζ.	Atom	x	у	Ζ.	Δx	Δy	Δz	$ \Delta $		
Sb11	0.93256	0.25	0.49808	Sb11'	0.93256	0.25	0.50192	0	0	0.0008	0.2732		
Sb18	0.44184	0.75	0.50272	Sb18'	0.44184	0.75	0.49728	0	0	0.0008	0.2732		
Cu20	0.68263	0.75	0.50526	Cu2'	0.68263	0.75	0.49474	0	0	0.0112	0.3825		
Cu2	0.18397	0.25	0.50594	Cu02'	0.18397	0.25	0.49406	0	0	0.0112	0.3825		



v c

















