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Morphological and chemical dynamics upon electrochemical cyclic sodiation of electrochromic tungsten oxide coatings extracted by *in situ* ellipsometry

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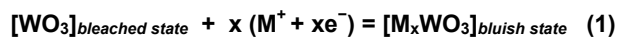
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The sodiation-desodiation process of sputtered amorphous electrochromic tungsten oxide coatings in aqueous-based medium was simultaneously monitored over 99 cycles by cyclic voltammetry and *in situ* spectroscopic ellipsometry. This allowed extracting the evolution of optical and geometrical parameters upon cycling. The resulting electrochemical coloring-bleaching process was dynamically fitted in the 1.8-2.8 eV optical range with a four-phase model including a constrained spline parametrization of the dielectric function. This allows real time access to thickness, surface roughness and dielectric function of Na_xWO₃. The temporal evolution of the latter in the fully colored state was used to highlight a porosity extent of the probed coating of opened morphology. The designed spectro-electrochemical approach was applied to map the temporal evolution of the Na content (x in Na_xWO₃) during and between cycles, taking into account the intricate interplay between charge density, thickness and electrolyte uptake. © 2020 Optical Society of America

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1. INTRODUCTION

Among other chromogenics used for sustainable energy, amorphous tungsten oxide (WO₃) is an extensively studied cathodic electrochromic (EC) material allowing a control of its optical properties in the visible and NIR ranges in response to both external electrical charge and electrolyte's ions, M, insertion: [1–3]



Corresponding EC devices cover a widespread set of eco-efficient smart windows [4,5]. If intercalations of Li, K or H ions are currently reported, they are not easily quantifiable (Li, H) and the case of hydrogen is also limited due to sensitivity to hygrometry. Moreover, K is seldom considered due to its poor ion conduction [6]. The intercalation of Na ions could then be alternatively considered despite its higher ionic radius, i.e. possible slower kinetics, compared to H and Li. In addition, the rising field of sodium-ion batteries is calling for a greater attention to the Na_xWO₃ bronze system as a

potential anode material [7]. Whatever the aforementioned applications, from the point of view of raw materials resources, solutions based on sodium appear more sustainable than those based on lithium [8].

Many studies focused on optical measurements and debated the coloration mechanism of sodium-tungsten bronzes. Thus, after pioneering works on optical properties [9], most reports on the dielectric functions were published in the 1970's and 1980's [10–15] whereas Tegg *et al.* computed it recently by means of DFT [16]. The deep blue cathodic coloration in such EC system [17] arises from the cumulative generation of electronic transitions between different Wⁿ⁺ states, but based on different mechanisms in the amorphous or crystalline states [1] i.e. localization (e.g. small polarons) or electron delocalization (Drude behavior), respectively. We showed recently [18] that optical absorption in amorphous Na_xWO₃ follows the generalized site saturation model as it was shown for Li_xWO₃ [19] or H_xWO₃ [20]. More particularly, we derived a unique optical model based on a constrained spline approach to draw out the pseudo-dielectric function spectra (0.6-4.8 eV) whatever the state of the EC layer, i.e., as-deposited, colored (x=0.1, 0.2 and 0.35), and bleached (x=0) by

means of *ex situ* spectroscopic ellipsometry (SE). This model allowed to assess the evolution of three polaronic transitions (*ca.* 1.3, 2.4, and 3.6 eV). Typically, all such EC systems reach a saturation level after the first coloring-bleaching cycles so that the amount of injected charges during coloring equals to that of extracted ones during bleaching. These preliminary cycles are considered as “activation” or “education” process [6,17]. The cationic intercalation is accompanied by a volume expansion of the oxide coating [5,18].

Another subject of interest is the reported degradation of EC performance of WO_3 resulting from the electrochemical intercalation in aqueous medium [21,22]. Amorphous WO_3 films dissolve in water to form tungstate-based ions which finally erode the films over several cycles i.e. a voltage-assisted dissolution [21]. The microstructure also plays an important role, the density of the films increasing their durability but increasing inversely their coloration and bleaching times [22]. Besides, if the phenomenon is slowed down in non-aqueous electrolyte thanks to both the poor ionizability and large size of organic molecules, together with the inhibition of tungstate ion release, this later can remain if the films are sufficiently hydrated [21,23]. Yet, data are still missing about the degradation for Na intercalation in aqueous medium. To the best of our knowledge, details on the evolution of the coating morphology during sodiation-desodiation cycle and throughout the whole electrochemical cycling are not available to date for this medium.

In this context, instead of transmission measurements [19,22,24] devoted to measure the optical modulation of coloring-bleaching process, optical methods based on reflective Spectroscopic Ellipsometry (SE) [25] are attractive alternatives as they bring more information about *in situ* processes, such as electrochemical ones occurring at the solid-liquid interfaces [26–28]. If significant researches have been conducted on WO_3 and M_xWO_3 systems by various groups, they essentially rely on *ex situ* monochromatic/spectroscopic or *in situ* monochromatic measurements [12,20,29–35]. Only a few ones have dealt with a full spectro-electrochemical approach [36,37]. For example, the open circuit dissolution of WO_3 films was monitored by ellipsometry in aqueous solution, pointing out a zero-order rate law whatever the aqueous media tested. However this study, performed from monochromatic ellipsometry, has only considered thickness variations without any hydration and/or roughening process due to the inherent limitations of the method. Witham et al. [36] extracted the pseudo-dielectric function (at 400 nm) during 3 cycles of hydrogenation-dehydrogenation process and modeled them using a Bruggeman effective medium approximation (BEMA) consisting of a mixture of fully colored and bleached states. They validated the model with the full spectral dependence (310-830 nm) during the coloration step. Camirand et al. obtained the *n* and *k* dispersion curves (250-950 nm) for H_xWO_3 ($0 < x < 0.32$) during 36 cycles of chronoamperometric hydrogenation-dehydrogenation process [37]. The authors accessed some related coating properties in every bleached state during the cycling, like roughness evolution or thickness loss. For these reasons, we opted to transpose *in situ* our above-mentioned constrained splines based optical model, i.e. in a continuous way during each electrochemical cycle between colored and bleached states (*x* between zero and maximum reachable values of *ca.* 0.35).

The purpose of this study is to investigate the ageing of Na_xWO_3 during electrochemical cyclic sodiation in aqueous medium. *In situ* spectroscopic ellipsometry (450-750 nm) was used in order to follow the chemical evolution during 99 cycles of sodiation-desodiation process and to gain insights into its durability.

2. METHODS

A. As-deposited samples

As-deposited samples consist of *ca.* $2.5 \times 2.5 \text{ cm}^2$ WO_3 reactively sputtered films on ITO-coated glass substrates (typically $\sim 500 \text{ nm}$ over 1 mm). The morphology of coatings was additionally imaged with field-emission scanning electron microscopy at 3 kV (FEG SEM; Gemini SEM 500, ZEISS, Germany) before and after the protocol described in the following

parts. Briefly, all of the as-deposited coatings were amorphous, sub-stoichiometric (typically $\text{WO}_{-2.7}$ but simply denoted as ‘ WO_3 ’). They possess an open morphology microstructure of columnar type suitable for electrochemical-intercalation of Na ions (Fig. 1a) thanks to a high working deposition pressure (4.5 Pa) [6]. The corresponding porosity ($\sim 30 \%$) has been previously determined by experimental density ($\rho = 4.72 \text{ g/cm}^3$) [18]. As aforementioned in the introduction, such morphology is known to accommodate a further volume change during the insertion-deinsertion of Na ions [5]. More details about the sputtering conditions can be found in [38].

B. Electrochemical cycling conditions

The control of the Na_xWO_3 optical properties is electrochemically performed via cyclic voltammetry (CV) using 0.1 M solution of Na_2SO_4 buffered by a phthalate salt at $\sim 2.7 \text{ pH}$ unit (0.1 M $\text{C}_8\text{H}_5\text{KO}_4 / 0.1 \text{ M HCl}$) as the electrolyte, saturated Ag/AgCl electrode as a reference and Pt counter electrode. The sample, as working electrode, was embedded in a dedicated electrochemical cell (see next part). Figure 1b illustrates the evolution of the cyclic voltammograms between the 1st and the 99th cycle. Each cycle consisted of a 25 mV/s cathodic scan from +0.45 V to -0.65 V, and then anodically back to +0.45 V. A pronounced evolution of the film morphology was observed during the experiment. It is evident by comparing Figs. 1a and 1c. Details of this evolution will be discussed in the results section.

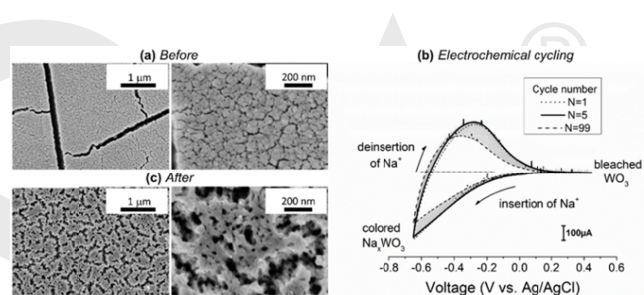


Fig. 1. (a) SEM plan view images for a representative sputtered amorphous WO_3 coating on ITO coated glass substrate before electrochemical cycling (b) First; 5th and 99th cyclic voltammograms of coloring-bleaching in a 0.1 M Na_2SO_4 solution buffered by a phthalate salt at $\sim 2.7 \text{ pH}$ unit. Electrode area was 1.37 cm^2 , and scan rate was 25 mV/s (c) Final SEM plan view images after electrochemical cycling.

C. In situ ellipsometry

1. Setup

Samples were analyzed *in situ* by spectroscopic ellipsometry (homemade setup [39]) during the aforementioned electrochemical cycling. Each sample was mounted as working electrode in a PolyEtherEtherKetone (PEEK) holder and placed vertically in a prototype coupling cell, which possesses two quartz windows allowing in and out polarized light beam. Prior to mounting each glass substrate back face was roughened in order to eliminate incoherent reflection. The incoming beam was emitted by a Xe-lamp (75 W) and the outgoing beam is split into two perpendicularly-polarized waves and simultaneously analyzed by two synchronized CCD detectors, providing final spectrum of averaged ellipsometric angles ($\Psi(\lambda), \Delta(\lambda)$), free of systematic errors. The angle of incidence was precisely determined around 65° . The EC system was here investigated between the bleached state and the colored one in a typical limit of 129 collected ellipsometric spectra for each cycle (i.e. *ca.* 1.5 spectra per second) for limiting both raw data volume and calculation effort. For the same reasons, the fitting was performed on a selection of 11 cycles among the 99 available. Each cycle dataset was assigned to an integer, *N*, defined as the cycle number (Fig. 1b).

2. Model and fitting procedure

In order to determine the optical constants of WO_3 dynamically during the sodiation-desodiation process, our previously mentioned model [18] is applied to kinetic SE data. Briefly, it consists of a four-phase structure i.e. electrolyte/surface rough layer/porous WO_3 layer/ITO coated-glass substrate. The optical constants of the substrate were previously determined with no EC layer and electrolyte is assimilated to water [40]. The top roughness was commonly treated as a BEMA of 50% WO_3 and 50% electrolyte [41]. For WO_3 , a first step consisted to extract its pseudo-dielectric function $\epsilon(\lambda)$. Subsequently, in a post-fit step, this effective EC layer is accordingly treated as an another BEMA mixture in order to compute a relative porosity f by the following equation:

$$0 = f \frac{\epsilon_{\text{H}_2\text{O}}(\lambda) - \epsilon(\lambda)}{\epsilon_{\text{H}_2\text{O}}(\lambda) + 2\epsilon(\lambda)} + (1 - f) \frac{\epsilon_{\text{ref}}(\lambda) - \epsilon(\lambda)}{\epsilon_{\text{ref}}(\lambda) + 2\epsilon(\lambda)} \quad (2)$$

All the pores in WO_3 are considered as fully filled by the electrolyte (dielectric function $\epsilon_{\text{H}_2\text{O}}$) with a f -electrolyte fraction and $(1-f)$ - WO_3 fraction (dielectric function ϵ_{ref}). ϵ_{ref} is relative to the reference 5th cycle in the fully colored state of WO_3 , as explained in section 3.A.1.

The fitting method is based on a home build constrained spline algorithm, which is Kramers Kronig consistent and free of all *a priori* on the dispersion of the dielectric function [42]. Extracted film properties of interest consist of both optical (pseudo-dielectric function) and morphological parameters (thickness, roughness, relative porosity). The fitting accuracy was expressed by the mean square error (MSE) between experimental and generated $(\Psi(\lambda), \Delta(\lambda))$ values, and checked to avoid ill-conditioning.

3. RESULTS AND DISCUSSION

A. Optical constants

1. Evolution during the initial reference cycle

The coatings turned alternatively in a colored state (bluish coloration) after applying the cathodic scan and in a bleached one upon reversing the potential during the anodic scan (Fig. 1b). Varying the potential in such a manner allows accessing continuously, through the optical model, to the pseudo-complex refractive index $\tilde{n} = n + ik$ or pseudo-dielectric function $\epsilon = \epsilon_1 + i\epsilon_2$ between the bleached and colored states. Figures 2a and 2b show the limits of evolution of \tilde{n} and corresponding ϵ during the first fitted cycle dataset. This sodiation-desodiation cycle is the 5th and is considered as the reference i.e. it is assumed that the oxide coating is conveniently activated and that degradation is neglecting. The corresponding full temporal evolution of ϵ_2 is also given in Video Clip 1 (see Visualization 1). After applying potential from 0.45 V to -0.65 V, n values decrease while k values increase at the same time in the covered part of visible spectrum. These trends are similar to those reported for H_xWO_3 bronzes [34,37]. Moreover pseudo-dielectric functions of Na_xWO_3 previously obtained *ex situ* under quite similar experimental conditions [18] are superimposed in Figure 2b for extended spectral range. One reasonably argue that *in situ*-determined ϵ_2 (Fig. 2b, gray line) can be qualitatively ascribed as a portion of the reported asymmetric polaronic absorption curve (Fig. 2b, black line). The latter is in fact the superimposition of the three contributions already mentioned in the introduction.

2. Evolution during consecutive electrochemical cycling

If the above-described behavior of the pseudo-optical constants remains the same after the initial reference cycle, the fully colored state becomes more and more deprecated from the 10th cycle, as depicted in terms of n and k in Fig. 2c and Fig. 2d. Indeed, inserts of Figures 2c and 2d show, at 633 nm, that n decreases from 1.7 to 1.5 while k increases from 0.28 to 0.11, corresponding to nearly linear degradation rates of 2.1×10^{-3} per cycle and 1.7×10^{-3} per cycle, namely 12% and 39%, respectively. The cause of such degradation can be found in the evolution of the coating morphology as emphasized hereinafter.

Figure 1c displays additional SEM surface pictures at the end of the electrochemical cycling, revealing a drastic extent of the coating's nanoporous network. This feature is found similar to aged WO_3 sol-gel coatings presenting an increase of channels crack size, suggesting a strong corrosion of the oxide network [23]. An excessive porosity level was also suggested by others to explain the degradation in long term durability of WO_3 coatings [22]. Such hypothesis was then tested to explain the evolution of pseudo-optical constants depicted in Figs. 2c-d, using Eq.(2) and plotted in Fig.3. After the 10th cycle, the electrolyte content already exceeds 3.8% and further increases up to ~50% plateau above the 90th cycle. Such extent of electrolyte penetration (swelling-like process [23]) is in qualitative agreement with the SEM plan views taken after the 99th cycle (Figs. 1a and 1c), as claimed [22,43].

To further validate the extracted dispersion curves, the coloration efficiency (CE) has been estimated as proposed by others [35,37]. This EC figure of merit represents the change in optical density (OD) per amount of charge density inserted ($[q_c]$) or deinserted ($[q_a]$) through the coating (Fig. 4, left axis). OD were here reversely simulated from the ellipsometric data for the selected cycles of Fig. 3. CE values at 650 nm upon cycling (Fig. 4, right axis) are in good accordance with the tabulated value for analogous insertion in 0.1 M Na_2SO_4 electrolyte (*ca.* $15 \text{ cm}^2 \cdot \text{C}^{-1}$) [44].

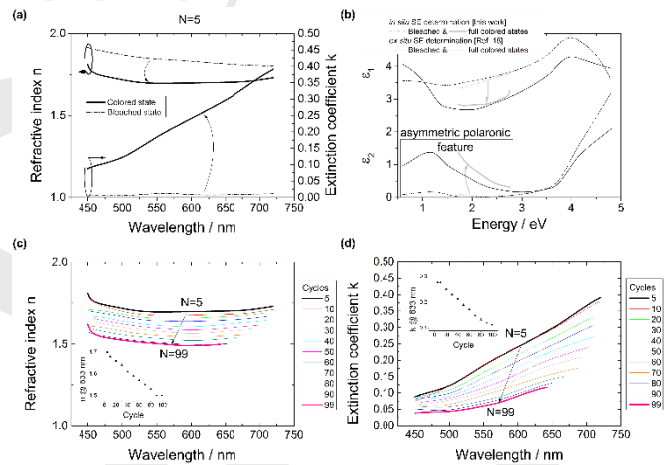


Fig.2 Best-fitted (a) pseudo-complex refractive index limits and (b) pseudo-dielectric function limits during the 5th cycle of sodiation-desodiation for a representative sputtered amorphous WO_3 coating, (c, d) Best-fitted pseudo-optical constants evolution associated with the fully colored state during electrochemical cycling. Inserts display the respective evolution of each optical constant at 633 nm. Additional Video Clips of ϵ_2 during each mentioned cycle are available as supplementary material (see Visualization 1 for N=5, Visualization 2 for N=10, Visualization 3 for N=20, Visualization 4 for N=30, Visualization 5 for N=40, Visualization 6 for N=50, Visualization 7 for N=60, Visualization 8 for N=70, Visualization 9 for N=80, Visualization 10 for N=90, and Visualization 11 for N=99).

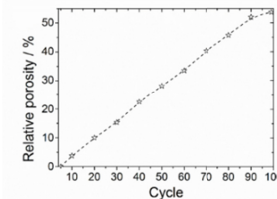


Fig. 3. Calculated porosity extent during the electrochemical cycling of Na_xWO_3 relatively to the reference 5th cycle. The dashed line is a guide to the eye.

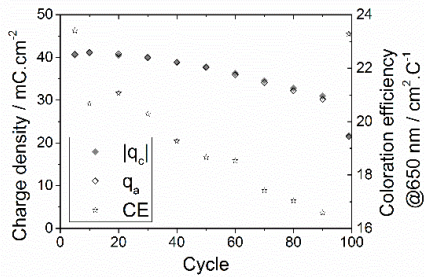


Fig. 4. (Left axis) Inserted (q_c) and extracted (q_a) charge densities and (Right axis) Coloration efficiency (CE) calculated at 650 nm for a representative sputtered amorphous WO_3 coating on ITO coated glass substrate, during the electrochemical cycling in a 0.1 M Na_2SO_4 solution buffered by a phthalate salt at ~ 2.7 pH unit.

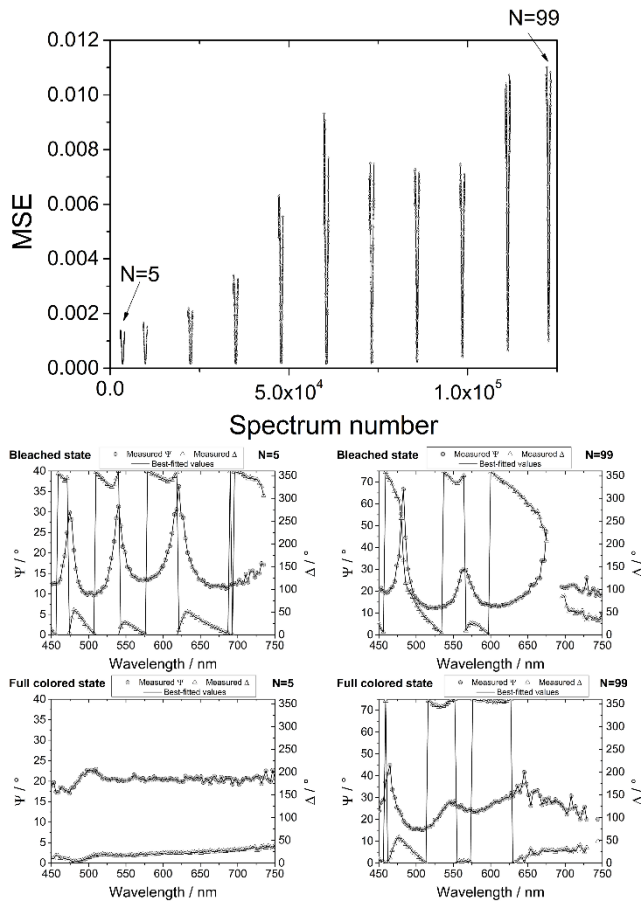


Fig. 5. (Top) Best-fitted mean square error (MSE) values during a whole sodiation-desodiation sequence for a representative sputtered amorphous WO_3 coating. To minimize the calculation effort only 11 cycle datasets on 99 available cycle datasets are fitted (each cycle dataset consists of *ca.* 129 spectra) (Bottom) Representative measured and best-fitted (Ψ, Δ) spectra for $N=5$ and $N=99$.

3. Fit quality

The top panel of Fig. 5 presents MSE values obtained during the whole sodiation-desodiation process. For the sake of brevity, only some cycles

between the 5th and the 99th cycle are represented. For each of the 11 cycle datasets, it is noticeable that a clear MSE minimum value exists and which strictly coincides to the maximum sodiation when correlated with corresponding electrochemical datasets, i.e. with the fully colored state. In contrast MSE maxima coincide with the bleached state. The overall MSE level (*ca.* ≤ 0.01 up to $N=99$) indicate that the output parameters can be considered with good confidence, i.e. the quality of the fit further supports the validity of the model. The bottom panel of Fig.5 displays some illustrative matches between measured and best-fitted spectra. However, the model better applies in the insertion state than in the deinsertion one. The relative drift of the MSE values from $N=5$ to $N=99$ is undoubtedly a sign of the ongoing deterioration of the coating as it has been explained above in Section 3.A.2, but with some 'optical healing' effect at the fully colored states (min. MSE *ca.* ≤ 0.001 up to $N=99$).

In addition, some misfits usually occur in more or less quantity due to the inherent quality of kinetic spectra that imply e.g. to reduce the fitted spectral range to get optical parameters, as shown in Fig. 2c and Fig. 2d. Yet, the general evolution remains valid.

B. Coulometric and geometrical factors

1. Charge density

After the first initial 5 cycles, the excessive porosity revealed by Fig. 3 impairs the sodiation-desodiation process in such a way that $|q_c| \neq q_a$ and CE values decrease with electrochemical cycling (Fig. 4). It is possible to get deeper in this analysis taking into account not only the charge during each cycle but also a full geometrical picture i.e. the available volume, in terms of porosity and both expansion and erosion-loss effects through thickness, to estimate the evolution of the Na content x in the tungsten bronze. This can be derived thanks to the dynamic properties determined by *in situ* SE.

2. Thickness and roughness

Based on observations of the film morphology before and after experiments (see Fig. 1) and knowing the dynamics of the electrical parameters, it is possible to derive a complete picture of the temporal evolution of both film geometry and composition.

Typical dynamics of the geometrical parameters are presented in Figure 6a for the reference cycle (i.e. the 5th). First of all, the thickness dynamics reveals the oxide expansion [38] upon Sodium insertion distinguishable from 0 to ~ 35 s, followed by the oxide contraction upon Sodium removal with another threshold from ~ 60 s to 88 s (Fig. 6a, filled symbols). In between, a mean level around 534 nm at 48 s can be estimated including the maximum Sodium insertion of this initial cycle (arrow on the axis of Fig. 6a). This thickness evolution obtained here *in situ* is consistent with previous *ex situ* one [18]. The roughness is also found to evolve in such a manner, the colored state being slightly rougher than the bleached one (Fig. 6a, half-filled symbols).

The morphological evolution during the 99 cycles is summarized in Figure 6b. The following quantities are calculated

$$d_{EXP} = \frac{d(CS,N) - d(BS,N)}{d(BS,N)} (\%) \quad (3)$$

$$rms_{EXP} = \frac{rms(BS,N) - rms(BS,5)}{rms(BS,5)} (\%) \quad (4)$$

where d_{EXP} and rms_{EXP} are the thickness expansion during a single cycle and the roughness expansion with respect to cycle $N=5$, respectively. $\overline{d}(i,N)$ and $\overline{rms}(j,N)$ are the mean values of the fitted thickness and roughness during the EC state i or j (maximum colored state quoted 'CS', or bleached state quoted 'BS') of the N^{th} cycle, respectively (see for example Fig. 6a, dashed lines, for $d(CS, 5)$ and $d(BS, 5)$).

First, the volume expansion trend is progressively strengthening up to a d_{EXP} value of *ca.* 14 %. Concerning the temporal evolution in the bleached state (not shown) it appears weak and too scarce to be considered as a marked effect. Secondly, the roughness also varies: the rms values in the

colored state are found slightly superior to these in the bleached state, whatever the cycle considered. Some analogous results using Chronoamperometric (CA) cycles in the case of H_xWO_3 in more severe electrolyte [37], but in the sole bleached state, reveal the following evolutions: roughness increases from 3 to 7 times after 36 CA cycles relative to as-deposited WO_3 and thickness losses of $\sim 2-10\%$ in the same time. In our case, there is no significant evolution of the thickness and roughness of the bleached between cycles. In conclusion, our chemical conditions are less acidic (2.7 instead of 1), and the potentiodynamic of CV leads to progressive cations insertion by comparison with fixed applied potential (CA).

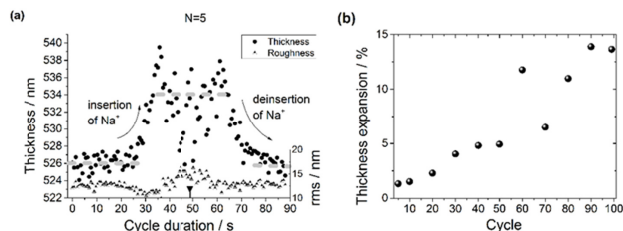


Fig. 6 (a) Best-fitted morphological parameters during the 5th cycle of sodiation-desodiation for a representative sputtered amorphous WO_3 coating. Arrow indicates the threshold of insertion-desinsertion of Sodium ions according Fig.1. The dashed lines are the mean values for the thickness in both limit EC states (b) Evolution of d_{EXP} during the electrochemical cycling.

3. Mapping of x

Finally, it is possible to derive a chemical information during the bronze degradation by estimating the evolution of the Na content, x (or Δx), through each fitted cycle datasets by applying usual assumptions [18]

$$x = \frac{n(Na)}{n(Na) + n(WO_3)} \quad (5)$$

where $n(Na)$ is the molar quantity of Sodium and $n(WO_3)$ this of tungsten oxide. The first quantity is calculated using electrochemical data

$$n(Na) = q_c(t) / F \quad (6)$$

where $q_c(t)$ is equal to the charge integration during insertion (Fig. 5) and F the Faraday constant. The second quantity is estimated as follows

For the initial reference cycle i.e. $N=5$: (reference for the further porosity extent)

$$n(WO_3) = S_a \times d(t) \times \rho / M \quad (7)$$

For the cycles $N/5 < N \leq 99$:

$$n(WO_3) = S_a \times d(t) \times (100 - f) \times \rho / M \quad (8)$$

where S_a is the active electrode area (1.37 cm^2), ρ the WO_3 density (see Section 2.A), M its molar mass (231.84 g/mol), f the porosity fraction determined for each cycle (Fig. 4) and $d(t)$ the thickness at each fitted time spectrum during the N^{th} cycle.

It is possible to map the evolution of x taking into account both volume expansion and loss (via d by ellipsometry) and the injected charge (via q by CV) during the whole electrochemical sequence. For a given cycle, the evolution of x vs time is a Gaussian-like curve (see insert of Fig. 7 for reference cycle $N=5$). The calculated dynamics during each cycle and with cycle number of x in Na_xWO_3 bronze coatings versus cycle duration and cycle is shown in Fig. 7. x shows a peak value slightly increasing up to the 90th cycle concomitantly as the pseudo-optical constants of Na_xWO_3 lower as more porosity occurs. Data of Fig. 6 evidence a change in film geometry upon sodium incorporation, in line with expansion of the molar volume already reported [4]. Moreover, increasing the number of cycles leads to progressive increase of the film porosity, leaving some free space for the molar volume expansion. Therefore, the maximal value of x reachable

during insertion is expected to increase with the number of cycles, as observed with our calculated values reported in Fig 7.

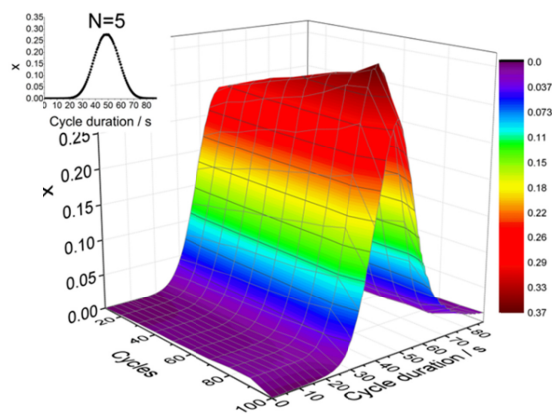


Fig. 7. Spectro-electrochemical determination of the Sodium content (x) evolution in Na_xWO_3 bronze coating, as function of electrochemical cycling. Insert: activated cycle ($N=5$)

In summary, the *in situ* spectro-electrochemical strategy developed enables deriving relevant information on both morphological and composition evolutions upon electrochemical cycling. It is found that as part of the tungsten oxide film is dissolved, it gains more ability to accommodate volume expansion during the cathodic phase and more sodium-ions can be incorporated per unit volume of the electrochromic film. This is in line with the fact that as-deposited films of opened morphology can more efficiently intercalate sodium ions than dense films [6,22].

4. CONCLUSION

In situ spectroscopic ellipsometry study was performed on half electrochromic cell operating in aqueous electrolyte. The voltage-driven sodiation-desodiation process of sub-stoichiometric sputtered amorphous tungsten oxide coatings was monitored to focus on the electrochemical cycling observed over 99 redox switches i.e. electro-coloring followed by electro-bleaching. Using a unique constrained spline parametrized model, both optical and morphological parameters were successfully fitted.

Analysis of the sodiation-desodiation process reveals a degradation after the first activation cycles. By considering the calculation of a coating porosity extent, the pseudo-dielectric functions are found in accordance with a progressive incorporation of more than half additional electrolyte content than in the initial cycles. This appears as the main reason for the pseudo-optical properties attenuation with time, whereas the corrected available volume for injection also provokes the sodium content to slightly increase.

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Disclosures

The authors declare no conflicts of interest.

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