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Co₂FeO₄@rGO composite: Towards trifunctional water splitting in alkaline media

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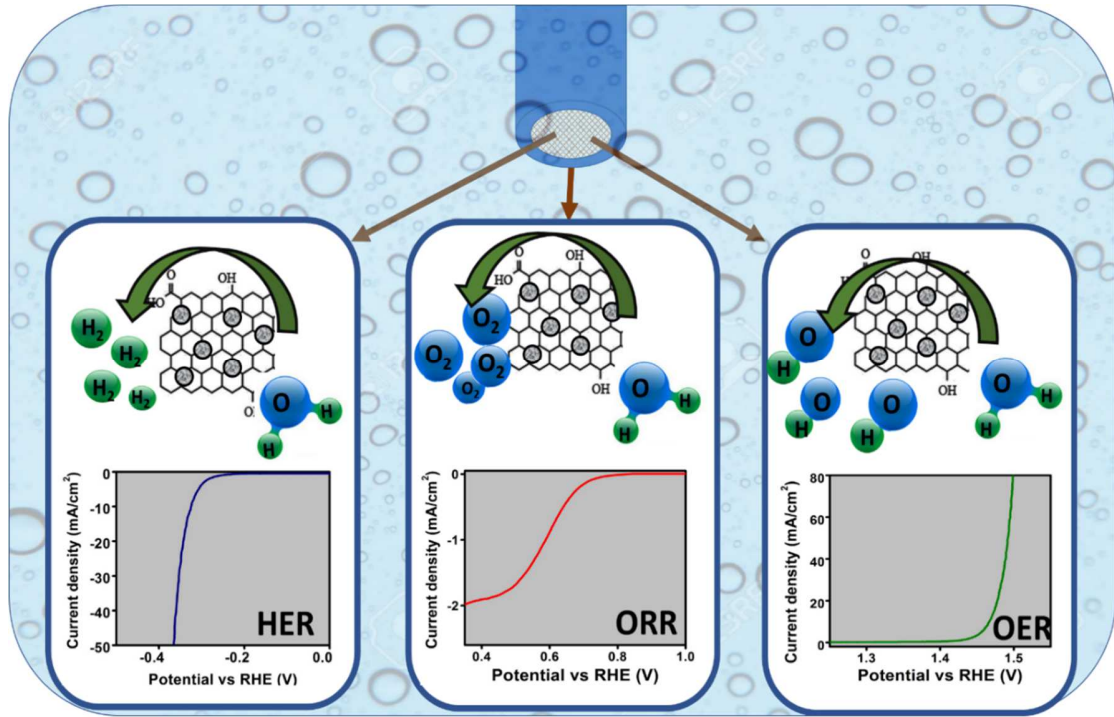
Abstract

Development of efficient, low cost and multifunctional electrocatalysts towards water splitting to harvest hydrogen fuels is a challenging task, but the combination of carbon materials with

transition metal-based compounds is providing unique and attractive strategy. Herein, composite systems based on cobalt ferrite oxide (Co_2FeO_4) @reduced graphene oxide (rGO) using simultaneous hydrothermal and chemical reduction method have been prepared. The proposed study eliminates one step of conversion of GO into rGO and it uses direct GO during the synthesis of cobalt ferrite oxide, consequently rGO based hybrid system is achieved with less labor and economy. The optimized composite has revealed an outstanding multifunctional application towards water splitting reactions such as oxygen reduction reaction (ORR), oxygen evolution reaction (OER) and hydrogen evolution reaction (HER). Various metal oxidation states and oxygen vacancies at the surface of Co_2FeO_4 @rGO composites guided the multifunctional surface properties. The optimized Co_2FeO_4 @ rGO composite presents excellent multifunctional properties with onset potential of 0.60 V for ORR, and an overpotential of 240 mV and 320 mV at a 20 mA cm^{-2} for OER and HER respectively. The outstanding multifunctionalities of the optimized Co_2FeO_4 @ rGO composite are associated to high electrical conductivity, high density of active sites, crystal defects, oxygen vacancies, and favorable electronic structure with the substitution of Fe for Co atoms in binary spinel oxide phase. These surface features synergistically uplifted the electrocatalytic properties of Co_2FeO_4 @rGO composites. The multifunctional properties of the Co_2FeO_4 @ rGO composite could be of high interest for its use in wide range of applications in sustainable and renewable energy fields.

Keywords: Reduced graphene oxide(rGO), bimetallic cobalt iron ferrite, composite systems, multifunctional electrocatalysts

Graphical Abstract: Illustration of multifunctional surface properties of $\text{Co}_2\text{FeO}_4@\text{rGO}$ composite



1. Introduction

Energy crisis and environmental pollution challenges require the development of outperforming energy conversion and storage devices [1, 2]. Water splitting for H₂ production is an appealing solution. Water splitting is able to transform solar energy for the generation of H₂ [3-7]. It is well established that the dynamic efficiency of oxygen reduction reaction (ORR), oxygen evolution reaction (OER) and hydrogen evolution reaction (HER) is crucial for the development of high performance metal air batteries, overall water splitting and hydrogen generation [8, 9, 10, 11]. The current challenge towards efficient water splitting is the immediate need of having outperforming nonprecious electrocatalysts. However, currently available electrocatalysts in the market are platinum (Pt) for the oxygen reduction reaction (ORR) as cathode in batteries, hydrogen evolution reaction (HER) as cathode and oxygen evolution reaction (OER) as anode in water electrolyzer, creating an economic issue due to their expensive nature [12]. To date, Pt and IrO₂/RuO₂ supported carbon are the-state-of-art electrocatalysts towards HER/ORR and OER, respectively.

The high cost and scarcity of Pt, Ir, Ru-based electrocatalysts for HER, ORR and OER create an economic stress, in addition to their poor electrochemical stability, therefore these obvious issues limit the large scale applications of precious electrocatalysts for water splitting [13]. Moreover, Pt/C could produce corrosion and decrease the physical force between the carbon support and Pt nanoparticles, which creates an aggregation of Pt nanoparticles and detachment from the carbon material. These are the worst aspects of using precious Pt/C electrocatalyst and its limitations towards large scale fuel cell commercialization [14]. This is the main reason to develop catalytic materials either with low content of Pt or completely Pt-free nonprecious catalysts to uplift the fuel cell technology [15]. Since two decades, noble metal-free electrocatalysts based on 3-d transition metals like (Fe, Co, Ni and Mo) and their oxides, sulfides, carbides, nitrides, phosphides and alloys have been investigated for simultaneous water splitting applications such as OER, ORR and HER due to their production from abundance resources, low cost and significant catalytic activity [16-21]. Additionally, these metal oxides have negligible aggregation during electrochemical testing in comparison to platinum or its compounds, therefore they are very favorable for the fuel cell technology [22]. The interfacial chemistry and density of catalytic sites of transition metal oxides can be enhanced via using substrates with high specific surface. The enhancement in the interfacial properties, high density of active sites and high surface to volume ratio are the demanding pillars towards the development of commercial fuel cells [23, 24], lithium ion battery [25], and

supercapacitors [26]. The binary spinel compounds have shown excellent catalytic activity [27, 28]. The use of Fe substituent in the binary spinel compounds has demonstrated an auxiliary and crucial role in improving the durability [29].

Among the binary spinel compounds, cobalt ferrite oxide exhibits a wide range of applications such as actuators, stress sensors, actuators, memory storage and magnetoelectric microwave devices and its magnetic properties are approximately equal to that of $Y_3Fe_5O_{12}$ ferrite [30, 31]. Cobalt ferrite oxide owing to its spinel structure offers plenty of attractive physical and chemical features. The spinel structure of cobalt ferrite oxide is carrying mixed oxidation states of A^{2+}/A^{3+} , and they further accelerate the kinetics of electrochemical reactions [32, 33]. It has been shown that the electronic structure of Co^{3+}_{Oct} in form of spinel Co_3O_4 could be tuned by addition of $Fe^{3+}_{Oct/Td}$. The replacement of Co atoms by Fe could induce the change in Co-O bond length and consequently effect on covalent character of bond [34].

Furthermore, the functionality, electrical conductivity, surface to volume ratio and stability of transition metal oxides could be highly increased through the addition of carbon based materials due to synergetic effect [35]. Wang et al. reported the bimetallic oxide of cobalt-iron (Co_2FeO_4) supported on multiwall carbon nanotubes as an electrocatalyst towards ORR process [36]. The conductive graphene oxide (GO) is a low cost carbon material and has promising potential towards lifting of catalytic properties [37]. Further reduction of GO into reduced graphene oxide (rGO) has higher electrical conductivity and it can have large possibility to enhance the surface area of metal oxides, thereby increase in current density and superfast charge transport is achievable [38]. In the rGO, the presence of π network is confirming the typical features of graphene [39, 40, 41]. The existing literature reveals that there are no reports on the cobalt ferrite oxide as multifunctional electrocatalyst towards three core HER, OER and ORR reactions. Also, there is no composite system of $Co_2FeO_4@rGO$ prepared by simultaneous reduction and hydrothermal method and its investigation as multifunctional electrocatalyst performing at the same time for HER, OER and ORR reactions. Furthermore, the variation in metal oxidation states and oxygen vacancies at the surface of $Co_2FeO_4@rGO$ composites together for the multifunctional electrochemical properties are not reported. To date

In this study, we have prepared hybrid $Co_2FeO_4@rGO$ systems by simultaneous hydrothermal and reduction method. The $Co_2FeO_4@rGO$ composite is characterized by a variety of analytical

techniques in terms of structure, composition, morphology, crystalline features, surface properties, stability and durability. The optimized $\text{Co}_2\text{FeO}_4@\text{rGO}$ composite has revealed multifunctional proprieties towards HER, OER and ORR reactions with pronounced performance.

2. Experimental work

2.1. Chemicals used:

Graphite, sodium Nitrate (NaNO_3), potassium permanganate (KMnO_4), hydrogen peroxide (H_2O_2), sulfuric acid (H_2SO_4), hydrochloric acid (HCl), cobalt chloride hexahydrate ($\text{CoCl}_2 \cdot 6\text{H}_2\text{O}$), iron chloride hexahydrate ($\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$), polyethylene glycol (PEG), 20%Pt/C, 20% RuO_2/C , potassium hydroxide (KOH) and hydrazine hydrate (N_2H_4). The chemicals were purchased from Sigma Aldrich, Karachi Pakistan. All the chemicals were of analytical grade and used without pretreatment. The precursors solutions were prepared in the deionized water.

2.2. Synthesis of $\text{Co}_2\text{FeO}_4@\text{rGO}$ composite nanostructures by simultaneous hydrothermal and reduction method

Desired nanostructures were synthesized by two steps. Initially graphene oxide (GO) was synthesized by modified Hummers method by taking 2 g of graphite flakes and NaNO_3 into H_2SO_4 in volumetric flask (1000 mL) under continuous stirring at approximately -5°C in ice bath. After the continuous stirring for 2 hours, 6 g of potassium permanganate was added carefully under the controlled conditions of temperature below 15°C . Then, ice bath was removed and the resultant solution was stirred at 35°C for 48 h to get brownish paste like material. Then, 100 mL of deionized water was added into it slowly. The reaction temperature was increased quickly up to 98°C and a brown color for the material was obtained. Furthermore, the solution was diluted with 200 mL of water under continuous stirring. The obtained solution was treated with 10 mL of H_2O_2 to terminate the mechanism to get a yellow color, then mixture was collected and washed with 10% of HCl and deionized water several times, and collected by centrifugation. Afterwards, the filtration and drying was performed under vacuum at room temperature, then a final product of GO powder was obtained [42, 43]. Secondly, 1M PEG, 0.02M cobalt chloride hexahydrate and 0.01M iron chloride were added into the deionized water and kept at magnetic stirring to get well dissolved precursor solution. Then, 1 mL of hydrazine was added drop wise into the solution under constant stirring at room temperature. Hydrazine is as reducing agent to induce GO to rGO transition and stabilizing

agent for the nanostructured materials. Later on, rGO was added with varying concentrations from 5 mg to 10 mg into the solution. The final solution was transferred into 100 mL Teflon autoclave with capacity of 70 mL and heated at 180°C for 5 hours in an electric oven. After the hydrothermal chemical reaction, as received precipitates were washed several times with the deionized water followed by ethanol. Eventually, the product was dried at 80°C in air overnight. Furthermore, the obtained product was treated at 500°C for two hours in air to get cobalt ferrite oxide@rGO as final product. The prepared Co_2FeO_4 @rGO composites with different quantities of 5 and 10 mg of rGO are referred to as CFG-5 and CFG-10. The pristine cobalt ferrite oxide sample was prepared via hydrothermal process without the addition of GO.

2.3. Physical characterization of different Co_2FeO_4 @rGO composite materials

The morphology of as prepared nanomaterials was studied through scanning electron microscopy (SEM) by JEOL JSM-6480A with operating voltage of 20kV. The energy dispersive spectroscopy (EDS) was applied to determine sample composition. Crystallographic study was carried out through X-ray diffraction (XRD) by using Rigaku TTR at 40 kV with high(?) intensity $\text{Cu-K}\alpha$ radiation ($\lambda=0.15406$ nm). Transmission electron microscopy (TEM) using a FEI Tecnai G2 (containing a Schottky emitter functioning at 200kV) was applied for further detailed studies of morphology. Chemical attachment of various elements was observed by X-ray photoelectron spectroscopy (XPS) by using PerkinElmer PHI 5100 XPS X-ray Photoelectron Spectrometer System. Curve fitting was performed using a combined Gaussian and Lorentzian line profile after Shirley's background subtraction by Casa XPS software. To analyze the chemical structure, the Raman spectroscopy was applied by using PerkinElmer RamanStation 400. Fourier transform infrared (FTIR) spectroscopy was applied (PerkinElmer Spectrum 2) on samples having shape of pellets to get information on the functional groups of various electrocatalysts in the region 4000 to 400 cm^{-1} . The UV-Vis spectroscopy by Lambda-365 was used to analyze absorbance of different nanocomposites which were dispersed in deionized water. X-ray photoelectron spectroscopy spectra were collected on a Kratos Axis Ultra (Kratos Analytical, UK) spectrometer equipped with a monochromatic $\text{Al K}\alpha$ (1486.6 eV). All spectra were recorded at a 90° takeoff angle. The high-resolution regions were acquired with 0.1 eV step and 20 eV pass energy (instrumental resolution better than 0.5 eV). Curve fitting was performed using a combined Gaussian and Lorentzian line profile after Shirley's background subtraction by Casa XPS software.

2.4. Electrochemical measurements of Co₂FeO₄@rGO composite materials

Different Electrochemical tests were carried out through an electrochemical workstation (Potentiostat VersaSTAT-4) equipped with a computer. For HER, OER and ORR all the experiments were done on three electrodes electrochemical system into 1.0 M KOH solution. Silver-silver chloride and graphite rod was used as reference electrode and counter electrode and modified glassy carbon electrode (GCE) was used as working electrode. To prepare catalyst ink for electrochemical tests, 5 mg of each catalyst material was added into 1 mL of deionized water and 0.5 ml of 5% Nafion as binder was added into the solution. The obtained catalyst solution was dispersed for 10 minutes in an ultrasonic bath. Later, 5 μ L of ink was coated on GCE by drop casting method and dried at room temperature. The loading mass of each catalyst was about 0.2 mg/cm². The catalytic activity towards HER and OER was studied through linear sweep voltammetry (LSV), under the potential window from 0 to -1.5V and 0 to 0.7 V at scan rate of 5 mV/s respectively. For ORR we have applied LSV at static conditions first, in the potential range from 0 to -1.2 V at various scan rates (5, 20, 40, 60, 80, 100 and 120 mV/sec). Secondly, ORR tests were carried by varying the rotation speed (0, 20, 40, 60, 80 rounds per minute, RPM) under fixed scan rate of 5 mV/sec. This RPM mechanism has been done through rotating assembly in 1.0 M KOH solution by magnetic stirring with fixed position of counter, reference and modified glassy carbon electrode. Electrochemical impedance spectroscopy (EIS) was done with sinusoidal potential of 10 mV and onset potential of 1.4 V versus reversible hydrogen electrode (RHE) at frequency range of 100 kHz to 0.1Hz. Chronopotentiometry was applied to investigate the durability of optimized composite system during OER and HER performance at 20, 40 and 60 mA/cm² and 10, 20 and 30 mA/cm² for 48 hours, respectively. The stability was studied before and after durability test through LSV technique. The double layer capacitance (C_{dl}) was calculated from CV curves in the voltage range from 0.05V to 0.25V at various scan rates (30, 50, 70, 90, 110 and 130 mV/s). Through Nernst equation, all given potentials are reported in this manuscript against RHE. The experimental part is briefly described in Supplementary scheme 1.

3. Results and Discussion

3.1. Morphology, crystalline structure, and chemical composition of various Co₂FeO₄@rGO nanostructured materials

The in situ transformation of GO into rGO during the cobalt ferrite synthesis has resulted a hybrid system which played a significant role in driving the water splitting at low energy consumption.

The XRD patterns of pristine GO, Co_2FeO_4 and various $\text{Co}_2\text{FeO}_4@\text{GO}$ composite materials are shown in Fig. 1. The diffraction patterns at 12.2° and 42.8° are correspond to (001) and (100) crystal planes respectively and these patterns demonstrate the typical characteristics of pristine GO [44]. There is some noisy background from 15-40o indicates the incomplete oxidation of graphite into GO, thus it is presented in the pristine GO sample. While, the pristine Co_2FeO_4 has various diffraction peaks at 19.1° , 31.3° , 36.7° , 44.9° , 59.5° and 65.1° assigned to (111), (220), (311), (400), (511), (440) crystal planes and they are fully agreed to reported work ³⁶. The addition of GO with different quantities during the synthesis of $\text{Co}_2\text{FeO}_4@\text{GO}$ composites brought substantial changes in the XRD patterns and intensities of reflections which is still unclear to us to identify in this study. It has been found the reflection of GO at (100) crystal plane was disappeared in the composites due to high probability of formation of thick layer Co_2FeO_4 crystalline on GO [45]. We have used a very small amount of GO in the hybrid system, whereas XRD measurement has mainly revealed the patterns of cobalt ferrite. However, XRD results have shown a minor shift in two theta angle towards low angle with the addition of GO and possibly it could create the defects in the composite material resulting to favorable catalytic activity. An addition of GO has also decreased the intensity of some of the cobalt ferrite oxide reflections during the composite formation and no another impurity was found during the XRD analysis.

Initially, the morphological features of pristine Co_2FeO_4 , GO and $\text{Co}_2\text{FeO}_4@\text{rGO}$ composites were analyzed through SEM equipped with EDS for the quantification of main elements. As clearly observed in Fig. 2(a), GO has exhibited a nanosheets like structure. The EDS spectrum of GO has confirmed the presence of C and O elements as shown in right side of Fig. 2(a). Co_2FeO_4 has revealed sphere like architecture as shown in Fig. 2(b). The EDS study has verified the presence of Co, Fe and O as main elements as shown in Fig. 2(b) and the obtained elemental information was found consistent with the published work [46]. The microflakes of GO are covered with oxide nanoparticles and remained underneath of cobalt ferrite oxide layer in the case of composite systems of CFG-5 and CFG-10 as shown in Fig. 2(c and d).

The $\text{Co}_2\text{FeO}_4@\text{rGO}$ composite system of CFG-5 has shown some irregular shape, whereas CFG-10 composite has shown highly intense nanoflakes like structure which could be attributed to excessive amount of GO. The presence of Fe, Co, O and C elements was quantified from $\text{Co}_2\text{FeO}_4@\text{rGO}$ composites through EDS analysis. The $\text{Co}_2\text{FeO}_4@\text{rGO}$ composites revealed a better surface features like small particle size in comparison to pristine GO and Co_2FeO_4 materials, thus they have shown better electrochemical properties. An addition of higher quantity of GO resulted in more carbon during the hydrothermal process as confirmed by EDS in terms of carbon content, which is further giving an idea of enhanced conductivity and large surface for the swift interaction of water molecules during electrolysis.

The HRTEM images and fast Fourier transform (FFT) patterns have been recorded to analyze the fine structure of various nanostructures at atomic level. Fig 3a shows HRTEM image of pristine Co_2FeO_4 which reveals well dispersed nanoparticles with different shapes and the corresponding FFT image is displayed at right side indicating the lattice fringes with d-spacing of 0.23 nm. Fig. 3b shows the nanosheets of graphene oxide and its FFT image, describing the typical d-spacing of 0.20 nm of graphene. The $\text{Co}_2\text{FeO}_4@\text{rGO}$ composites (Fig. 3c and d), characterized with slight folding at the edges correlated to reduced graphene oxide layers. The FFT images of both $\text{Co}_2\text{FeO}_4@\text{rGO}$ composites exhibited the lattice fringes with d-spacing of 0.24 nm and 0.25 nm for CFG-5 and CFG-10 composites, respectively. The presence of plenty edges in the $\text{Co}_2\text{FeO}_4@\text{rGO}$ composites are providing significant catalytic sites for the water catalysis. The lattice fringes of the composite systems are comparable to those of pristine rGO and cobalt ferrite oxide, suggesting a high possibility of chemical coupling between GO and cobalt ferrite oxide. The strong chemical coupling is further enhancing the hybrid material stability and reproducibility during the electrochemical characterization.

The chemical composition and the valence state of GO, Co_2FeO_4 , CFG-5 and CFG-10 were investigated by XPS. The corresponding XPS survey scans show that the prepared samples including the hybrid nanomaterials are of high purity (Figure S1 a). C1s features show that GO is in a reduced form in the $\text{Co}_2\text{FeO}_4@\text{rGO}$ composites since the contributions related to the carbon-oxygen bonds are diminished compared to GO (Figure S1 b). Figure shows O1s, Fe2p and Co2p XPS features of CFG-5 and CFG-10. The O1s peak was de-convoluted into two peaks positioned at 530.2-530.3 (O_I) and 531.4-531.6 eV (O_{II}), corresponding to the stoichiometric oxidized metal-

oxygen bonds and oxygen vacancies, respectively [47]. The O_{II} area proportion ($O_{II}/(O_{II}+O_I)$) represents the relative quantity of oxygen vacancy. CFG-5 and CFG-10 exhibit 53 and 51 % of oxygen vacancies, respectively. Fe 2p_{3/2} was deconvoluted into Fe²⁺ (710.60-710.70 eV) and Fe³⁺ (712.5-713.00 eV) peaks [48] as shown in Figures 4c and 4d. Fe²⁺/Fe³⁺ ratio is calculated to be 0.23 and 0.62 for CFG-5 and CFG-10, respectively. Co 2p_{3/2} doublet was well fitted by the tetrahedral Co³⁺ (780 eV) and octahedral Co²⁺ (781.3 eV) contributions [49, 50]. Co³⁺/Co²⁺ ratio is found to be ~ 0.44 and 0.70 for CFG-5 and CFG-10, respectively. The XPS analysis has confirmed the higher concentrations of Fe²⁺ in CFG-5, Co³⁺ in CFG-10 and oxygen vacancies in CFG-5, suggesting that an addition of GO has dominant role in altering the metal oxidation states and oxygen vacancies at the surface of Co₂FeO₄@rGO composites, further driving the multifunctional surface properties.

The Raman spectroscopy has been applied on various nanostructures to analyze their band states. As shown in supplementary Fig. S2(a), three corresponding peaks have been analyzed, the D band at 1353 cm⁻¹, G band at 1580 cm⁻¹ and the 2D band at around 2705 cm⁻¹ in the Raman spectrum of carbon atoms in graphene oxide sheets. The G band corresponds to sp² hybridization of carbon atoms, the D band is commonly related to sp³ carbon defects [51, 52]. As for Co₂FeO₄, all the relevant bands can be clearly observed at 298 cm⁻¹, 476 cm⁻¹ and 680 cm⁻¹ which correspond to the presence of Co, Fe and O elements as depicted in supplementary Fig. S2(b) [53]. The Raman shifts for the Co₂FeO₄@rGO composites exhibit slight changes in the corresponding band intensity as shown in supplementary Fig. S2(c). Interestingly, the addition of graphene oxide has decreased intensity of D band and slight enhancement of G band is observed. As expected an addition of hydrazine in precursor of cobalt ferrite oxide resulted to the reduction of GO into rGO as confirmed from decrease in the intensity of D band [54]. Raman and XPS results have successfully verified that we have achieved the in-situ growth of cobalt iron ferrite and rGO composite in the single step. The Raman spectroscopy analysis has shown strong interaction between GO and Co₂FeO₄ in terms of band position and enhancement of G band intensity, indicating another evidence for strong chemical coupling of GO with cobalt ferrite oxide in the hybrid system. The Raman study is supporting the observations made on the electrochemical reactions related to the enhanced conductivity and multifunctional properties.

The FTIR analysis was performed to identify the functional groups within pristine GO, Co₂FeO₄ and Co₂FeO₄@rGO composites. Generally, the absorption bands of O-H, C-H, C=O, O-H, C-O

and C=C groups were located at 3440, 2930, 1790, 1025, and 873 cm^{-1} , respectively as depicted in supplementary Fig. S2(a). The metal oxide (M-O) absorption bands were seen at 650 and 590 cm^{-1} which further suggest the presence of typical metal oxide bonds like Co-O and Fe-O bonds [55]. As shown in supplementary, the broader absorption bands are shown due to nanostructure phase of cobalt ferrite oxide. Moreover, the addition of GO has made significant change in stretching vibration bands of pristine Co_2FeO_4 sample in the hybrid system, which could be related to the interaction between binary metal oxide and GO as shown in supplementary Figure S3 (a). Furthermore, the UV-visible absorption study was carried on various nanostructures to confirm optical features of as prepared materials. As shown in Fig. S3(b), the absorption peak of pristine Co_2FeO_4 observed at 280 nm which corresponds to Co-O existence within the composite [56, 57]. The pristine GO has absorbance peak at 232 nm which corresponds to π - π^* transition of remaining sp^2 C=C bonds. However, as prepared Co_2FeO_4 @rGO composites including CFG-5 and CFG-10 have shown slight change in the absorption intensity at 232 nm compare to pure Co_2FeO_4 .

3.2. Multifunctional perspectives of Co_2FeO_4 @rGO composite as an electrocatalyst

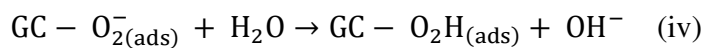
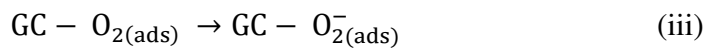
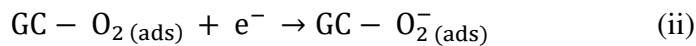
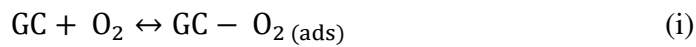
The multifunctional properties of in situ produced Co_2FeO_4 @rGO composites and their pristine components were studied toward ORR, OER and HER characterization.

It has been identified that ORR as a backbone reaction of fuel cells and metal-air batteries [58, 59]. For this purpose, we tested the different as prepared nanostructures including pristine GO, Co_2FeO_4 , composite Co_2FeO_4 @rGO materials and 20%Pt/C. The evaluation of ORR functionality of these materials in terms of half wave potential was studied by linear sweep voltammetry in oxygenated 1.0 M KOH aqueous solution in a potential window of 0.2 to 1.12 at sweeping rate of 5 mVs^{-1} as shown in Fig. 5(a). The corresponding half wave potential values of 0.59, 0.58, 0.56 and 0.60 V were observed for pristine GO, and Co_2FeO_4 , and Co_2FeO_4 @rGO composites including CFG-5 and CFG-10 respectively and in comparison to the state-of-art Pt/C catalyst with half wave potential of 0.85 V as shown in Fig. 5(b). The obtained results confirm that the Co_2FeO_4 @rGO composite like CFG-10 has superior ORR activity as compared with previous reported research work as shown in Table (S3). The corresponding K-L plots of CFG-10 are depicted in Fig. 5(c). Note that the great linearity along with consistent slope has been observed. The linearity of K-L plots shows the first order reaction kinetics followed by the Co_2FeO_4 @rGO composite like CFG-10 sample with increasing concentration of dissolved oxygen into electrolytic solution [58, 59]. It

is obvious from the excellent linearity of K-L plots that Co₂FeO₄@rGO composite like CFG-10 has high probability for constant supply of electrons due to its uniform distribution onto the GC electrode, indicating the typical catalytic performance of the composite material.

Furthermore, the effect of rotation per minutes (RPM) at various rates such as 0, 200, 400, 600, and 800 was investigated on ORR activity at a scan rate of 5 mVs⁻¹ as shown in Figure 6. There was a linear relationship built with the current density and various values of RPM due to the short distance of electron transport channel between the electrolyte and electrode interfaces designated to variety of active centers available for ORR reaction [60]. The RPM study has revealed an excellent ORR activity by the Co₂FeO₄@rGO composite like CFG-10 sample. Additionally, the effect of various applied potentials on the RPM for understanding the ORR activity of various nanostructured materials was studied as shown in Fig. 6(e). The ORR activity of different nanostructured materials was found static under the used scan rates of 5, 20, 40, 60, 80, 100 and 120 mVs⁻¹ as shown as Fig. (S4). This observation has confirmed significant improvement in the ORR activity of Co₂FeO₄@rGO composite like CFG-10 sample.

A general approach has been built to describe ORR reaction mechanism on the carbon based materials involving several steps. The initial step, the adsorption and electron transport take place to produce superoxide (O₂⁻). In next step, either the reduction or disproportionation of superoxide is expected. Under alkaline conditions, two waves are clearly found in ORR polarization curves connected to oxygen reduction or peroxide species taking place at various active centers on the surface of carbon based material [61, 62]. The ORR reaction mechanism on the modified GC electrode is generally followed by reported work [63].



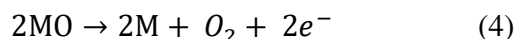
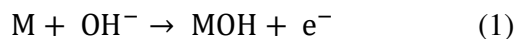
The reactant and product described in reaction (iii) are two different forms of superoxide ion on the surface of carbon material [64-66].

Another aspect of water splitting is the OER reaction and it has been considered a very critical for the cathode reaction of rechargeable metal-air batteries and as anode in the overall splitting process [67]. The OER is found kinetically slow half-cell reaction in water splitting as it involves four electron transfer process, therefore an efficient electrocatalyst has to be developed to overcome this challenge in order to exploit complete renewable energy from water splitting.

The OER LSV curves were measured for various synthesized nanostructured materials such as pristine GO, pure cobalt ferrite oxide, 20%RuO₂/C, Co₂FeO₄@rGO composites like CFG-5 and CFG-10 at a scan rate of 5 mVs⁻¹ in 1.0 M KOH aqueous solution. The LSV curves of pristine GO and Co₂FeO₄ have shown relatively poor OER activity with an onset potential of 1.58V and 1.52 V vs RHE. The Co₂FeO₄@rGO composites like CFG-5 and CFG-10 revealed a dramatic decrease in OER onset potential of 1.49V and 1.47V vs RHE respectively as shown in Fig 7(a). The measured overpotentials of pristine GO, pure Co₂FeO₄, Co₂FeO₄@rGO composites like CFG-5 and CFG-10 at 20 mAcm⁻² are 350, 290, 260, and 240 mV and these overpotential values have been extracted from the theoretical thermodynamic potential of 1.23 V for water splitting. For better pictorial presentation, the overpotentials of prepared materials are enclosed in Fig. 7(e), highlighting the superior performance of Co₂FeO₄@rGO composite like CFG-10 sample. The OER performance of Co₂FeO₄@rGO composite like CFG-10 in terms of low overpotential was compared with recently published results on nonprecious electrocatalysts supplementary Table (S2). The composite has superior or comparable performance to those excellent developed electrocatalysts. The enhanced performance of Co₂FeO₄@rGO composites could be attributed to several factors such as synergetic effect between cobalt ferrite oxide and GO through active sites from Co₂FeO₄ and electrical conductivity from GO, defects in the structure including oxygen vacancies, morphology of composite materials, large ECSA value, fast charge transport value and an excellent interfacial chemistry between the well chemically anchored cobalt ferrite oxide onto GO [68, 69]. The Tafel analysis was used to study the OER kinetics using Tafel equation as shown in Fig. 7(b). The calculated Tafel slopes are 105, 68, 56 and 51 mVdec⁻¹ for the pristine GO, pure Co₂FeO₄, Co₂FeO₄@rGO composites like CFG-5 and CFG-10 respectively. The Tafel study has

revealed an outstanding OER kinetics on the Co₂FeO₄@rGO composites from practical point of view compare to the noble metal RuO₂ based catalysts.

The OER reaction processes via four electron transfer mechanism, generally on the surface of metal oxides under alkaline conditions is described as follows:



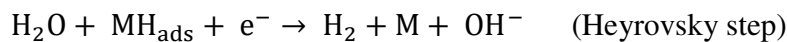
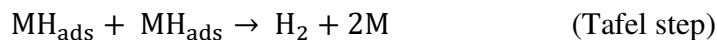
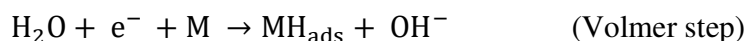
Theoretically, Tafel values are assumed on the basis of above four steps as 120, 60, 40, and 15 mVdec⁻¹. In our case, the rate limiting step of OER kinetics is governed by step 2.

The durability and stability are very essential to investigate for the nonprecious electrocatalysts as generally they are limited by these issues for long term applications. This is the reason, we studied the durability of Co₂FeO₄@rGO composite like CFG-10 using chronopotentiometry test at various current densities of 20, 40, 60 mA/cm² for 48 hours as shown in Fig. 7(c). These results have shown an excellent durable response of this composite system confirming its potential long term applications. These outstanding durability performance is possibly related to strong chemical coupling of cobalt ferrite oxide onto GO leading to provision of multichannel for charge transport without any abrupt changes. The stability was studied by the comparative analysis of LSV curves before and after the durability experiment as shown in Fig. 7(d), resulting no shift in the onset potential, current density and overpotential. Therefore, outperform stability is an excellent indicator for Co₂FeO₄@rGO composite like CFG-10 towards long term and large scale applications. To understand the scientific reason during OER process, EIS study was used to quantify the charge transfer resistance value of various nanostructured materials at OER onset potential and 1.0 M KOH solution. The EIS data was analyzed by two ways including Bode plots and Nyquist plots. The Bode plots are enclosed in in supplementary data Fig. (S6) and corresponding Nyquist plots are depicted in Fig. 7(f). The corresponding values from EIS calculation are given in Table 1. The Co₂FeO₄@GO composite like CFG-10 has shown superior phase angle as compare to CFG-5 and both pristine GO and cobalt ferrite oxide materials. The

charge transfer resistance (R_{ct}) values were 1020, 780.8, 395 and 58.5 Ω for GO, pristine Co_2FeO_4 , Co_2FeO_4 @rGO composites like CFG-5 and CFG-10 respectively. The low charge transfer resistance of Co_2FeO_4 @rGO composite like CFG-10 is again verifying an excellent charge transfer between modified electrode and electrolyte. Hence, accelerated OER kinetics has been demonstrated on the surface of Co_2FeO_4 @rGO composite like CFG-10 sample.

In addition to the excellent bi-functional ORR and OER activities, Co_2FeO_4 @rGO composite like CFG-10 also demonstrate a significant activity as cathode material towards HER in water splitting. The LSV polarization curves were measured for GO, pristine Co_2FeO_4 , Co_2FeO_4 @rGO composites like CFG-5 and CFG-10 in 1.0 M KOH at scan rate of 5 mVs^{-1} as shown in Fig. 8(a). The HER performance of these nonprecious nanostructured materials was evaluated with respect to noble metal 20%Pt/C. The measured overpotential of GO, pristine Co_2FeO_4 , Co_2FeO_4 @rGO composites like CFG-5 and CFG-10 are 470, 420, 380, and 320 at 10 mAcm^{-2} . These high values of overpotential suggest that pristine GO and cobalt ferrite oxide have poor HER activity due to few catalytic active centers, consequently limited electrochemical performance is observed. However, the Co_2FeO_4 @rGO composites like CFG-5 and CFG-10 have relatively low overpotential and indicating the role of composite material in terms of synergetic effect toward enhanced HER performance. The Co_2FeO_4 @rGO composite like CFG-10 has shown superior or comparable performance to those recently reported HER nonprecious electrocatalysts in alkaline conditions as given in Table (S1). This superior performance owned by Co_2FeO_4 @rGO composite like CFG-10 is attributed to high density active sites, electrical conductivity, swift electron transfer and favorable surface properties [70, 71]. For better readability and visibility, the measured overpotential of various presented materials are shown in Figure 8(e). The HER kinetics was evaluated in terms of Tafel analysis using Tafel equation and the obtained results are enclosed in Fig. 8(b). The Tafel slopes of GO, pristine Co_2FeO_4 , Co_2FeO_4 @rGO composites like CFG-5 and CFG-10 were measured as 152, 80, 59 and 48 mVdec^{-1} respectively. The Co_2FeO_4 @rGO composites like CFG-10 has exhibited low Tafel slope value signifying the faster HER kinetics via swift two electron transfer and closely related to that of 20%Pt/C. The chronopotentiometry was used to investigate the durability of Co_2FeO_4 @rGO composites like CFG-10 during HER process at different current densities of 10, 20 and 30 mAcm^{-2} . The durability results for 48 hours confirmed the outperform performance for lone term applications as depicted in Fig. 8(c). There was no flocculation in potential drop during durability measurement due to strong chemical

bonding between GO and cobalt ferrite oxide resulting to static and fast charge transport during HER reaction. Furthermore, the stability was analyzed through LSV curves behavior before and after the durability measurement as shown in Figure 8(d). The stability results were found satisfactory without any abrupt variation in the onset potential and current density during HER reaction. It has been shown that charge transport during electrochemical HER reaction is essential to determine as it is directly linked to the kinetics of reaction kinetics. Therefore, we have carried out EIS in 1M KOH and the obtained Nyquist plots for the description of charge transfer are shown in Fig. 8(f). However, the Bode plots are enclosed in supplementary Fig. (S6). The charge transfer values of GO, pristine Co_2FeO_4 , Co_2FeO_4 @rGO composites like CFG-5 and CFG-10 were observed as with corresponding R_{ct} values of 1230, 960.5, 380.75 and 64.8 Ω respectively. The Co_2FeO_4 , Co_2FeO_4 @rGO composite like CFG-10 exhibited a fast charge transport properties owing to the presence of highly conductive host substrate GO for cobalt ferrite oxide nanostructures, thus revealing a favorable HER kinetics. Under alkaline conditions, the HER kinetics expressed by three step as described in equations (1), (2), and (3) as given below:



The first step is related to Volmer step, which involves the adsorption of protons onto the electrode surface as indicated in equation (1), then followed by the recombination of adsorbed intermediates as represented in equation (2) or Heyrovsky step, which involves the sorption of the adsorbed species as expressed in equation (3) [72]. The HER mechanism on Co_2FeO_4 @rGO composite like CFG-10 is followed by Heyrovsky step.

The evaluated high performance electrocatalytic properties of Co_2FeO_4 @rGO composite like CFG-10 could be attributed from the following factors. 1. The simultaneous hydrothermal and reduction of has offered high electrical conductive material rGO with unique structure for hosting the cobalt ferrite oxide nanostructures and multi electron transfer channels [73-75]. 2. The use of Fe substituent for the Co enabled the superfast durability of composite material, it eliminates the aggregation of Co during electrochemical reactions, with exposure of optimum catalytic sites, and electrochemical stability [76]. 3. The strong chemical coupling of binary metal oxide with rGO

further excited the composite material to behave efficiently towards ORR, OER and HER leading to its multifunctional applications [77, 78, 79]. The variation in the metal oxidation states and oxygen vacancies at the surface of Co₂FeO₄@rGO composites due to varying amount of GO is responsible for the multifunctional surface properties. 5. The use of in situ cobalt ferrite and rGO process favored the three demanding functionalities of water splitting at the surface of composite system. Moreover, based on the physical characterizations, the improved electrocatalytic multifunctional applications are attributed to defects in the crystal structure of composite as indicated by XRD, small particle size surface and large quantity of rGO of composite provided the favorable surface for frequent interaction of water molecules during electrolysis as revealed from SEM and EDS analysis, HRTEM confirmed the favorable catalytic edges, strong coupling between rGO and cobalt ferrite oxide enable the composite to allow the steady and continuous charge transport without the compromise on the catalytic performance during water splitting, and the presence of variant metallic oxidation states. Additionally, the excellent electrocatalytic activities of Co₂FeO₄@rGO composite like CFG-10 are compared with recently reported multifunctional electrocatalysts (Table S4) confirming as developed Co₂FeO₄@rGO composite like CFG-10 would be an excellent choice in terms of low overpotential, high current density and low cost as an alternative multifunctional electrocatalyst employed in the practical applications of water splitting.

To get deeper insight on the improved multifunctional applications of GO, pristine Co₂FeO₄, Co₂FeO₄@rGO composites like CFG-5 and CFG-10, electrochemically active surface area (ECSA) and double layer capacitance were calculated from the non-faradic region of CV curves at various sweeping scans. The CV curves were recorded against different scan rates such as 30, 50, 70, 90, 110, 130 mVs⁻¹ as shown in supplementary Fig. (S5). The C_{dl} values were calculated from CV curves through linear fitting of current density versus different scan rates as depicted in Fig. S5(e). The C_{dl} values have been calculated as 3.1, 7.3, 16.6 and 23.2 μF/cm² for GO, pristine Co₂FeO₄, Co₂FeO₄@rGO composites like CFG-5 and CFG-10 respectively. The ECSA has been calculated through C_{dl} values through following relation;

$$ECSA = C_{dl}/C_s$$

In the given expression, C_{dl} is double layer capacitance and C_s is specific capacitance at the interface of electrode ($C_s=0.04 \mu\text{F}/\text{cm}^2$) for KOH medium. The calculated ECSA values are 77.5, 182.5, 415 and 580 cm^2 for GO, pristine Co_2FeO_4 , $\text{Co}_2\text{FeO}_4@\text{rGO}$ composites like CFG-5 and CFG-10 respectively as shown in Table 1. The obtained values of ECSA suggest that, addition of GO at optimum level has dramatically enhanced the active sites of Co_2FeO_4 and they have shown vital role in overall electrochemical performance towards core OER, HER and ORR reactions. The change in the morphology and lattice fringes after the durability test of OER and HER were evaluated by SEM, EDS and HRTEM as shown in S7 and S8. This analysis is suggesting a negligible change in the shape and crystal deformation revealing an outstanding stability of composite material for long term applications.

4. Conclusions

In this study, we have used simultaneous hydrothermal and reduction approach for the synthesis of various composite materials based on cobalt ferrite oxide and reduced graphene oxide. The direct use of GO was intentionally changed into rGO during the preparation of cobalt ferrite oxide nanostructures based hybrid system. The synthesis strategy avoids the use of one step of GO into rGO conversion and provides two step synthesis for the hybrid material based on cobalt ferrite oxide@rGO. The $\text{Co}_2\text{FeO}_4@\text{rGO}$ composite like CFG-10 has demonstrated excellent multifunctional applications towards ORR, OER and HER, the core reactions in water splitting. The composite material has shown remarkable comparable or superior to that noble metal catalysts like Pt/C and RuO₂/C owing to its high surface area with exposure of optimum catalytic sites, high electrical conductivity and multiple charge transport channels. The $\text{Co}_2\text{FeO}_4@\text{rGO}$ composite like CFG-10 has exhibited onset potential of 0.60 for ORR, and an overpotential of 240 mV and 320 mV for OER and HER respectively in alkaline media. The chronopotentiometry test at various current densities revealed an outstanding durability for the period of 48 hours suggesting the long term applications of proposed multifunctional electrocatalyst based on $\text{Co}_2\text{FeO}_4@\text{rGO}$ composite like CFG-10. The superior performance of composite is attributed to high active surface area of 580 cm^2 , defects in the structure, oxygen vacancies, fast charge transport and large exposure of active sites of cobalt ferrite oxide on the surface of reduced graphene oxide. Potentially and

promisingly the advance strategy for the design of multifunctional nonprecious electrocatalysts would be an alternative roadmap to harvest hydrogen fuels from electrochemical water splitting.

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Data Availability Statement

Data available in article supplementary material

5. References

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Figure Captions

Fig. 1 X-ray diffraction patterns of GO, Co₂FeO₄, CFG-5 composite and CFG-10 composite

Fig. 2 SEM images of various nanostructures with relevant EDS spectra (a) GO (b) Co₂FeO₄ (c) CFG-5 composite and (d) CFG-10 composite

Fig. 3 HRTEM images of various nanostructures and lattice fringes with corresponding FFT pattern (a) Co₂FeO₄ (b) GO (c) CFG-5 composite (d) CFG-10 composite

Fig. 4 XPS spectra for the O1s signal of (a) CFG-5 composite and (b) CFG-10 composite, for the Fe2p signal of (c) CFG-5 composite and (d) CFG-10 composite and the Co2p signal of (e) CFG-5 composite and (f) CFG-10 composite

Fig. 5 (a) LSV curves for ORR activity in 1M KOH solution at 800 RPM with reference of Pt/C (b) Calculated half wave potential values for each nanostructure (c) The Koutecky-Levich (K-L) plots for CFG-10 composite at various potentials

Fig. 6 LSV curves for ORR performance of various nanostructures at 5 mV/s for different RPMs and potentials respectively (a) GO (b) Co₂FeO₄ (c) CFG-5 composite (d) CFG-10 composite and (e) Corresponding linear fitting of current density vs RPM for ORR activity of each nanostructure at fixed 0.4 V potential

Fig. 7 (a) LSV curves of various nanostructures like Co₂FeO₄, GO, Co₂FeO₄@rGO composites (CFG-5 and CFG-10), 20%RuO₂/C at 5 mV/s for OER performance (b) Corresponding Tafel plots (c) Chronopotentiometry durability of Co₂FeO₄@rGO composite (CFG-10), at different current density values of 20, 40 and 60 mA/cm² (d) Stability before and after durability test of Co₂FeO₄@rGO composite (CFG-10), (e) Histogram for overpotential values of Co₂FeO₄, GO, Co₂FeO₄@rGO composites (CFG-5 and CFG-10), (f) Nyquist plots for OER performance through EIS study of Co₂FeO₄, GO, Co₂FeO₄@rGO composites (CFG-5 and CFG-10)

Fig. 8 (a) LSV curves of various materials like Co₂FeO₄, GO, Co₂FeO₄@rGO composites (CFG-5 and CFG-10), 20%Pt/C at 5 mV/s for HER performance (b) Corresponding Tafel plots (c) Chronopotentiometry durability of Co₂FeO₄@rGO composites (CFG-10), at 10, 20 and 30 mA/cm² (d) stability before and after durability of Co₂FeO₄@rGO composite (CFG-10), (e) Histogram for overpotential values of Co₂FeO₄, GO, Co₂FeO₄@rGO composites (CFG-5 and CFG-10), (f) Nyquist plots for HER performance through EIS study of Co₂FeO₄, GO, Co₂FeO₄@rGO composites (CFG-5 and CFG-10)

Fig. 1

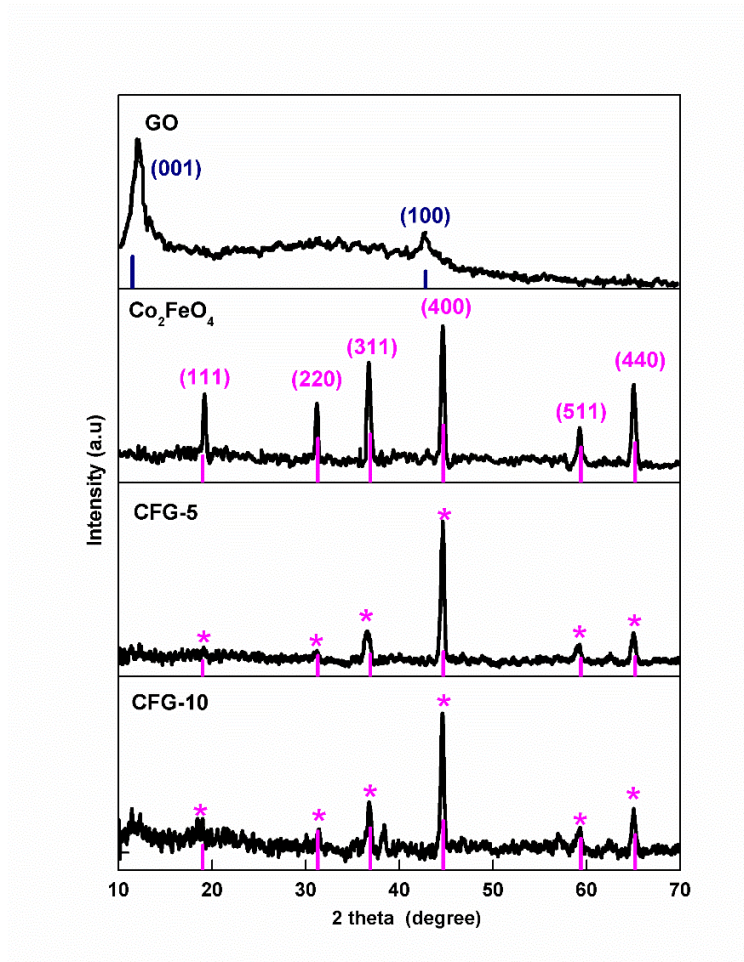


Fig. 2

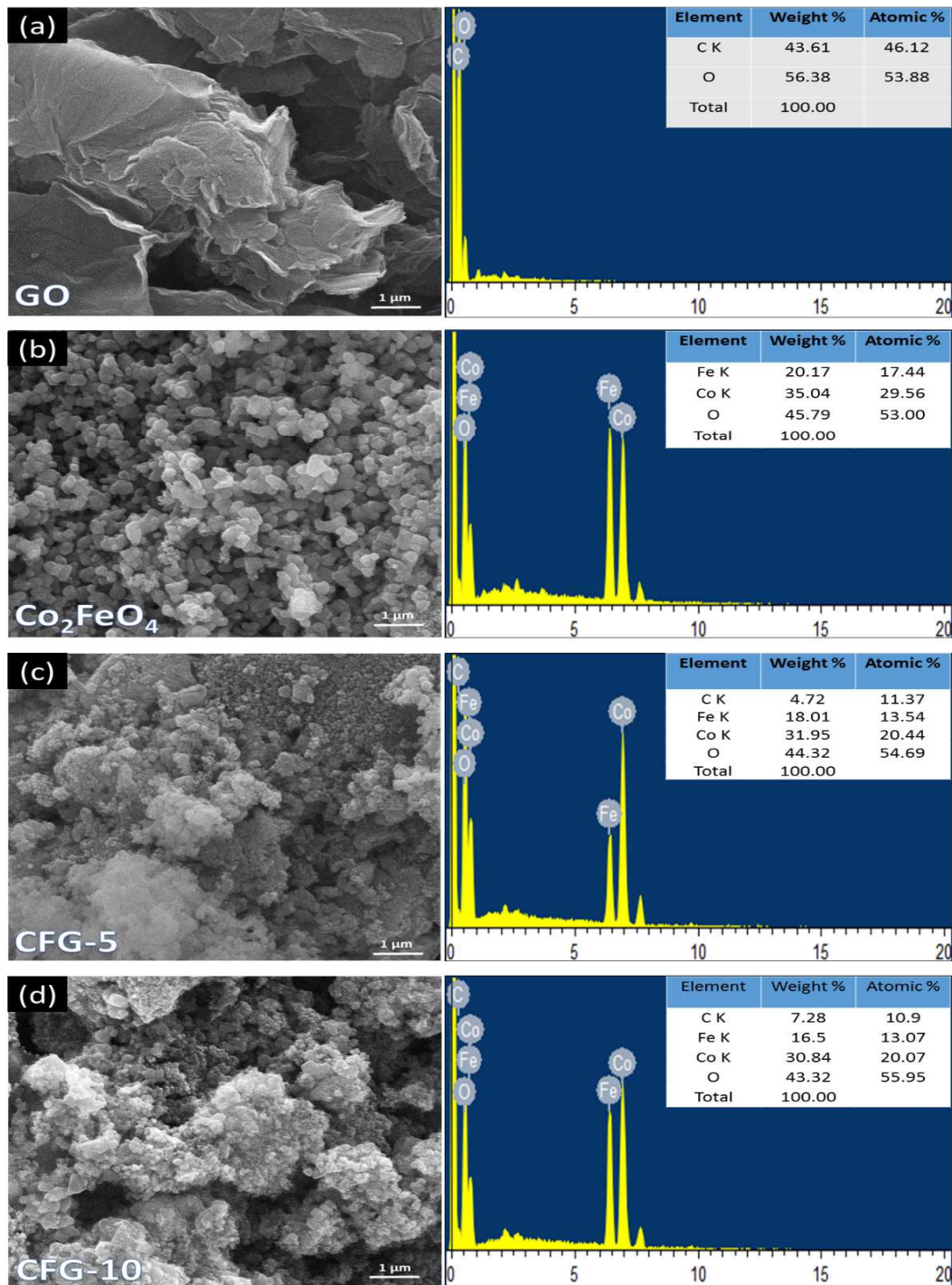


Fig. 3

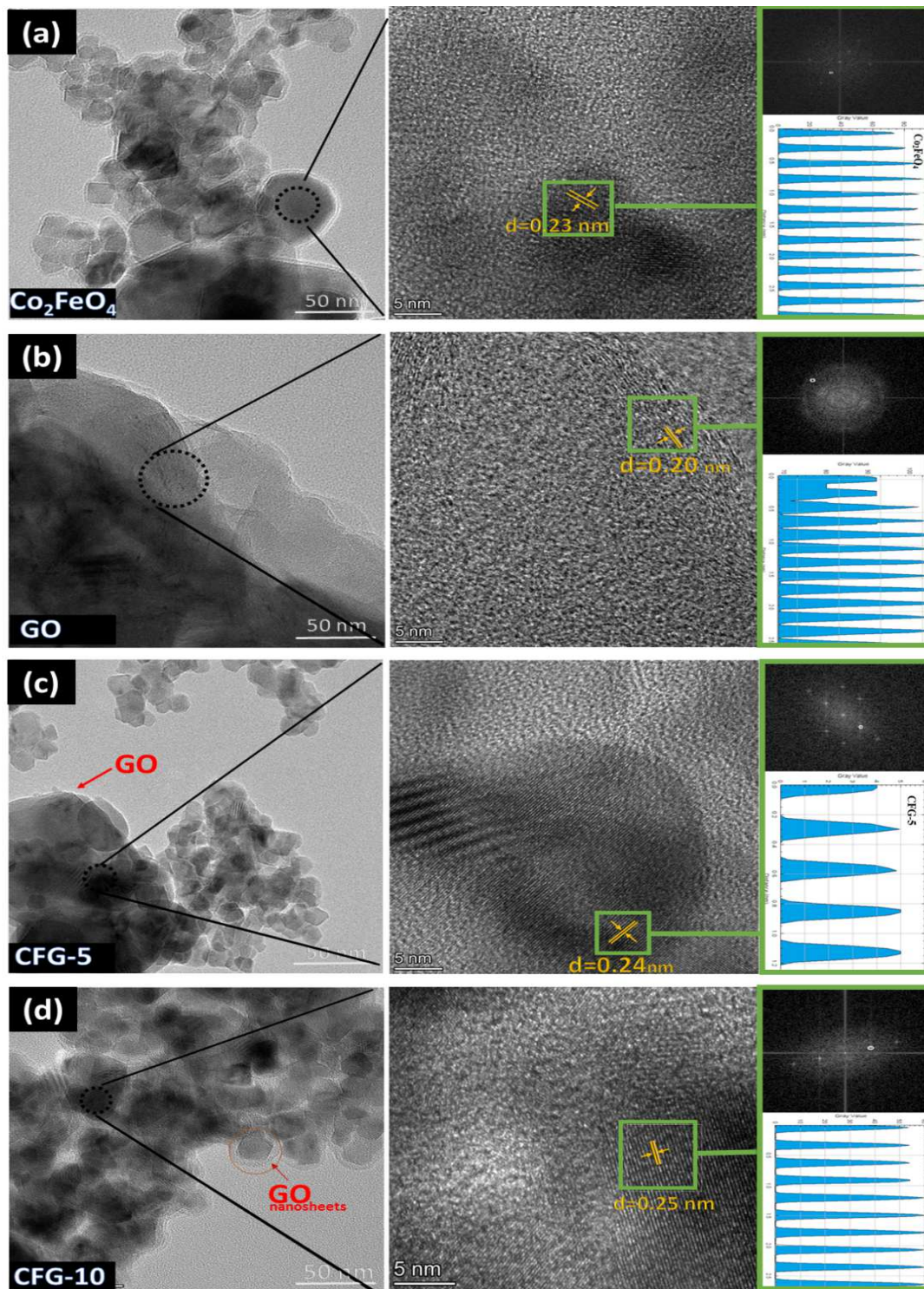


Fig. 4

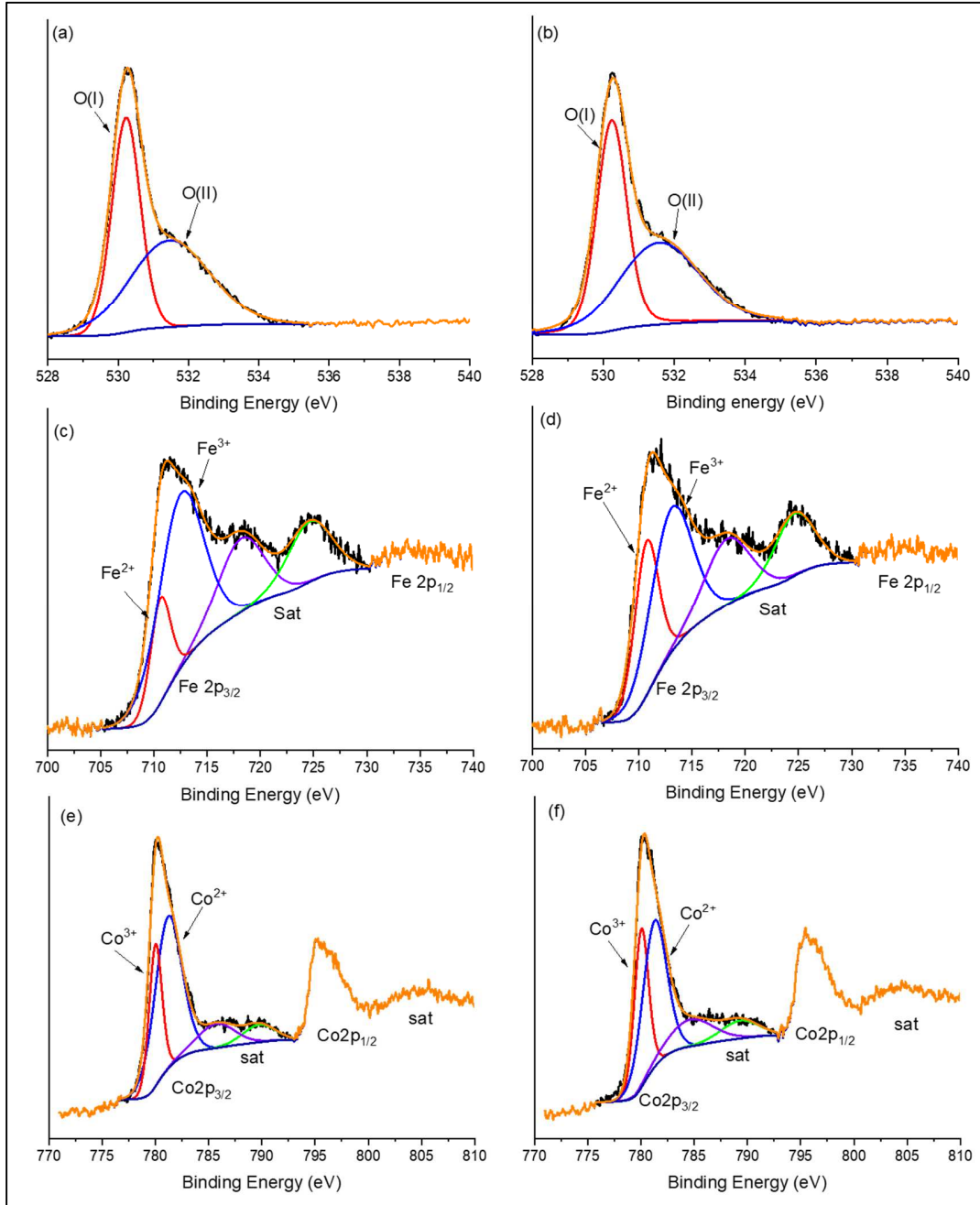


Fig. 5

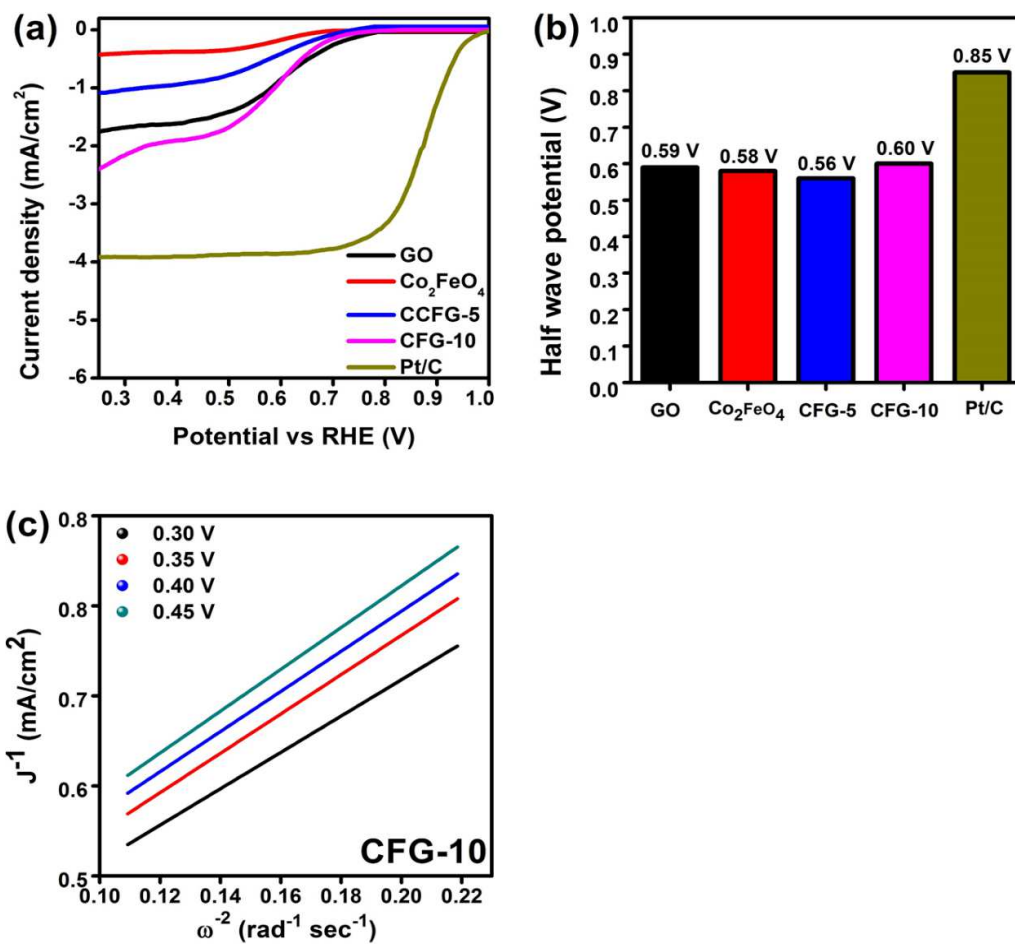


Fig. 6

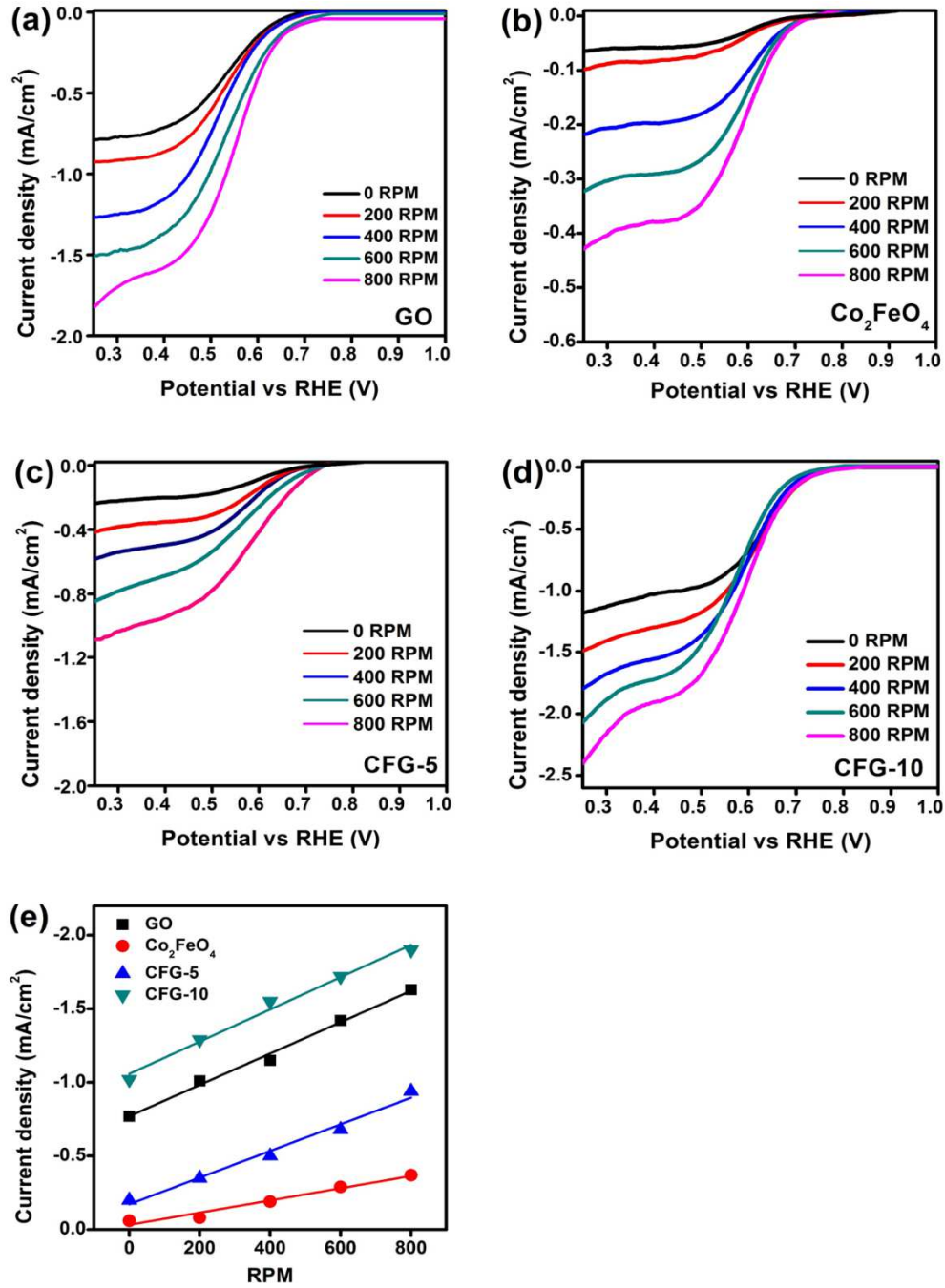


Fig. 7

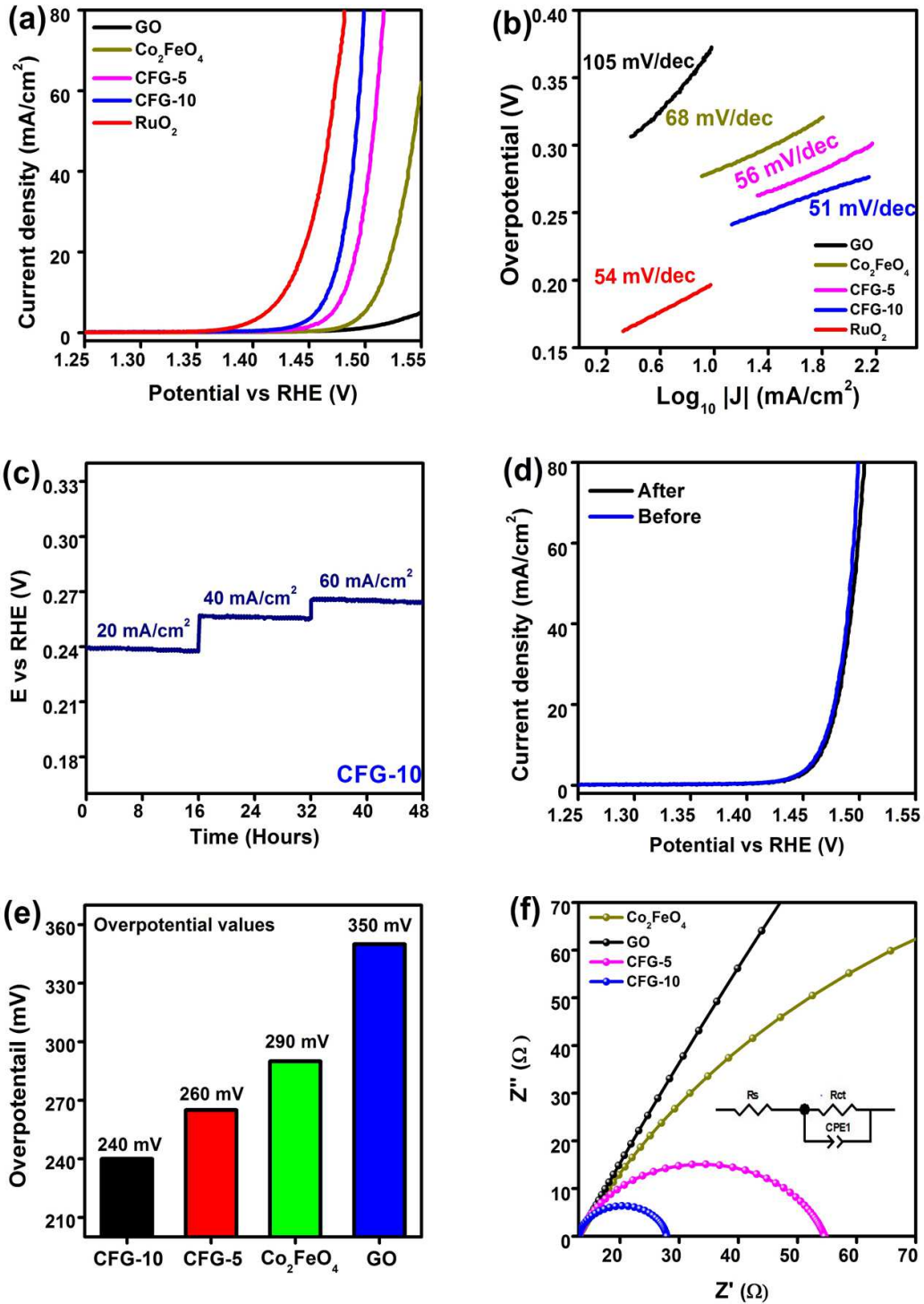


Fig. 8

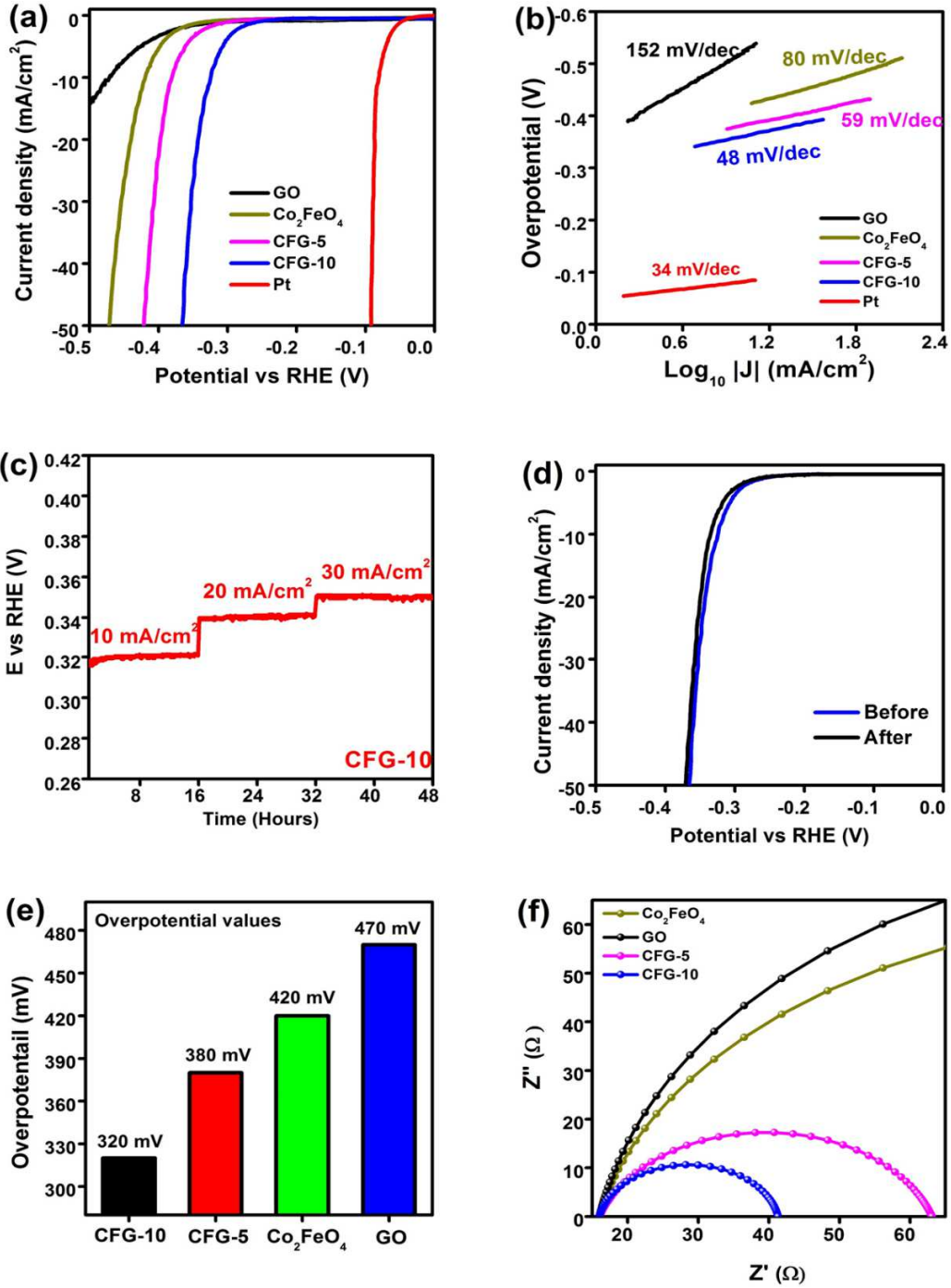


Table 1. Electrochemical features of various electrocatalysts for OER, HER and ORR

| <u>Catalyst</u> | <u>Calculation by LSV</u> | | | | | | <u>Calculation by EIS</u> | | | | <u>Calculation by CV</u> | |
|---|---------------------------|--|---------------|--|-----------------|---------------------|----------------------------|--------------------------|----------------------------|--------------------------|--------------------------|-------------------------------------|
| | OER | | HER | | ORR | | OER | | HER | | Double Layer Capacitance | Electrochemical Active Surface Area |
| | Tafel Slope | Overpotential at 20 mA/cm ² | Tafel Slope | Overpotential at 10 mA/cm ² | Onset Potential | Half wave Potential | Charge Transfer Resistance | Double Layer Capacitance | Charge Transfer Resistance | Double Layer Capacitance | | |
| | <i>B</i> | η | <i>B</i> | η | <i>V vs RHE</i> | <i>V vs RHE</i> | <i>R_{ct}</i> | <i>CPE_{dl}</i> | <i>R_{ct}</i> | <i>CPE_{dl}</i> | <i>C_{dl}</i> | <i>ECSA</i> |
| | <i>mV/dec</i> | <i>mV</i> | <i>mV/dec</i> | <i>mV</i> | <i>V</i> | <i>V</i> | Ω | <i>mF</i> | Ω | <i>mF</i> | $\mu F/cm^2$ | <i>cm^2</i> |
| GO | 105 | 350 | 152 | 470 | 0.74 | 0.59 | 1020 | 0.32 | 1230 | 0.29 | 3.1 | 77.5 |
| Pristine Co₂FeO₄ | 68 | 290 | 80 | 420 | 0.66 | 0.58 | 780.8 | 1.31 | 960.5 | 1.45 | 7.3 | 182.5 |
| CFG-5 | 56 | 260 | 59 | 380 | 0.69 | 0.56 | 395 | 2.72 | 380.7 | 2.10 | 16.6 | 415 |
| CFG-10 | 51 | 240 | 48 | 320 | 0.72 | 0.60 | 58.5 | 5.24 | 64.8 | 4.95 | 23.2 | 580 |

Supporting Information

Co₂FeO₄@rGO composite: Towards trifunctional water splitting in alkaline media

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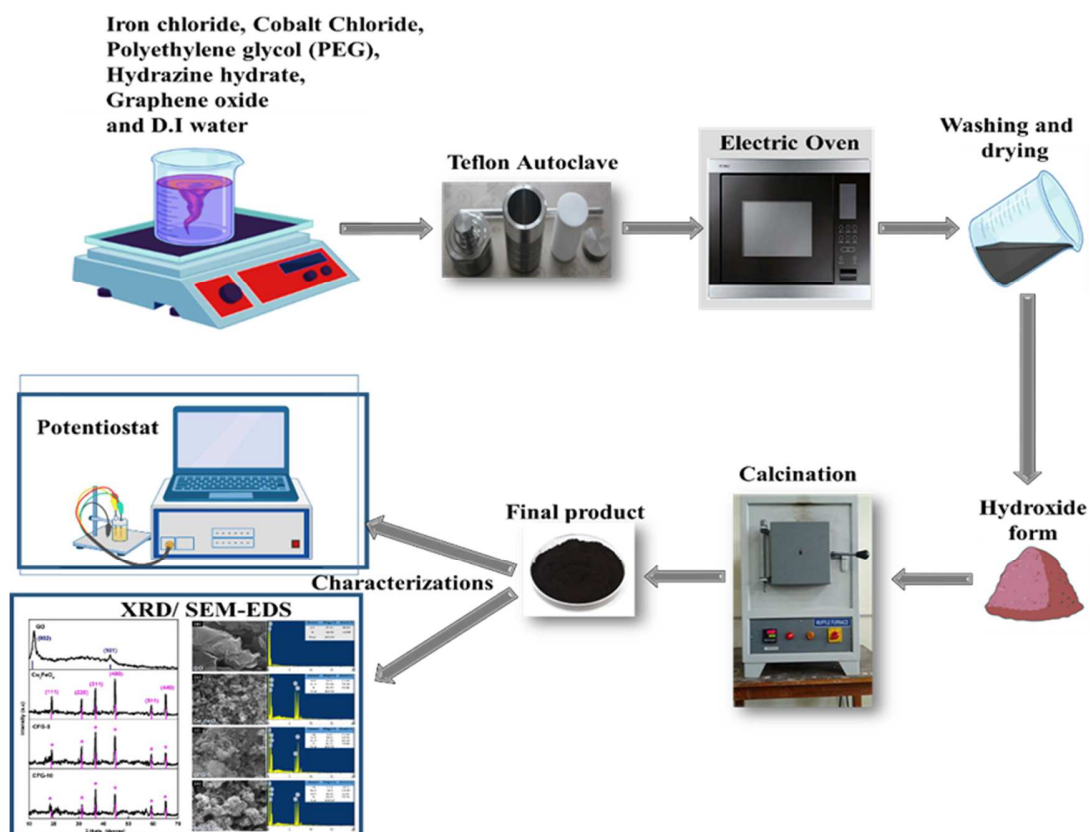
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Scheme 1: Synthesis process of $\text{Co}_2\text{FeO}_4@\text{rGO}$ composite, structural characterization and electrochemical set up for multifunctional surface properties

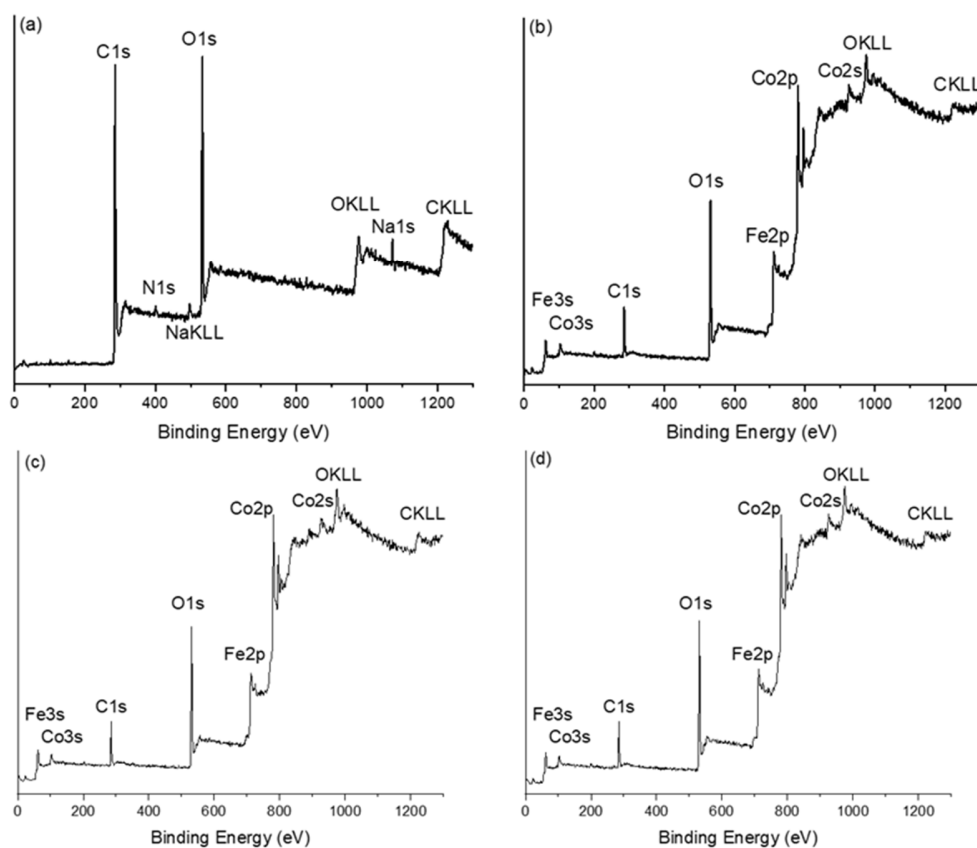


Fig S1(a) XPS Survey scan (a) GO, (b) Co_2FeO_4 , (c) CFG-5, (d) CFG-10

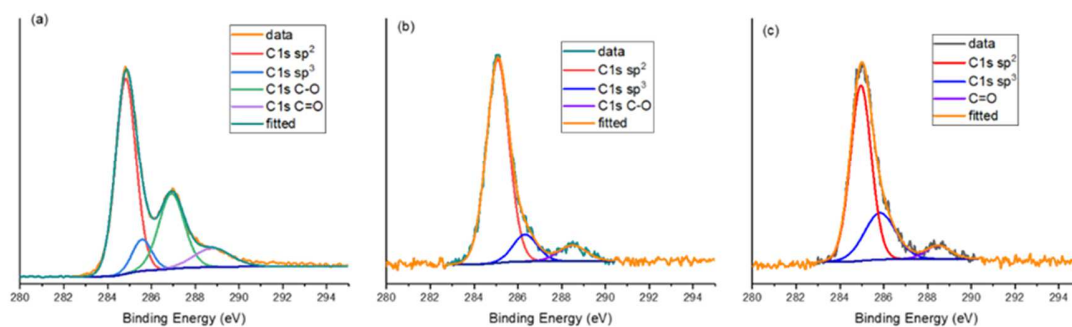


Fig. S1(b) XPS C1s (a) GO, (b) CFG-5, (c) CFG-10

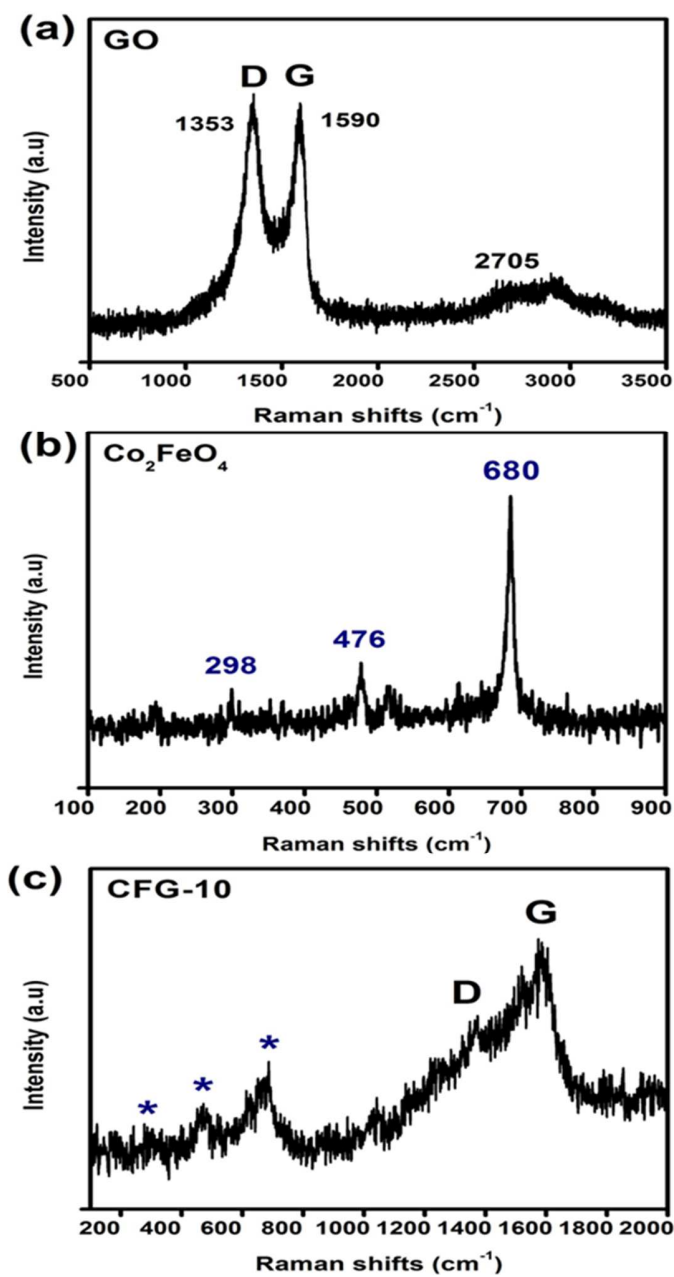


Fig. S2 Raman spectroscopy of various nanostructures (a) GO (b) Co₂FeO₄ and (c) CFG-10 composite

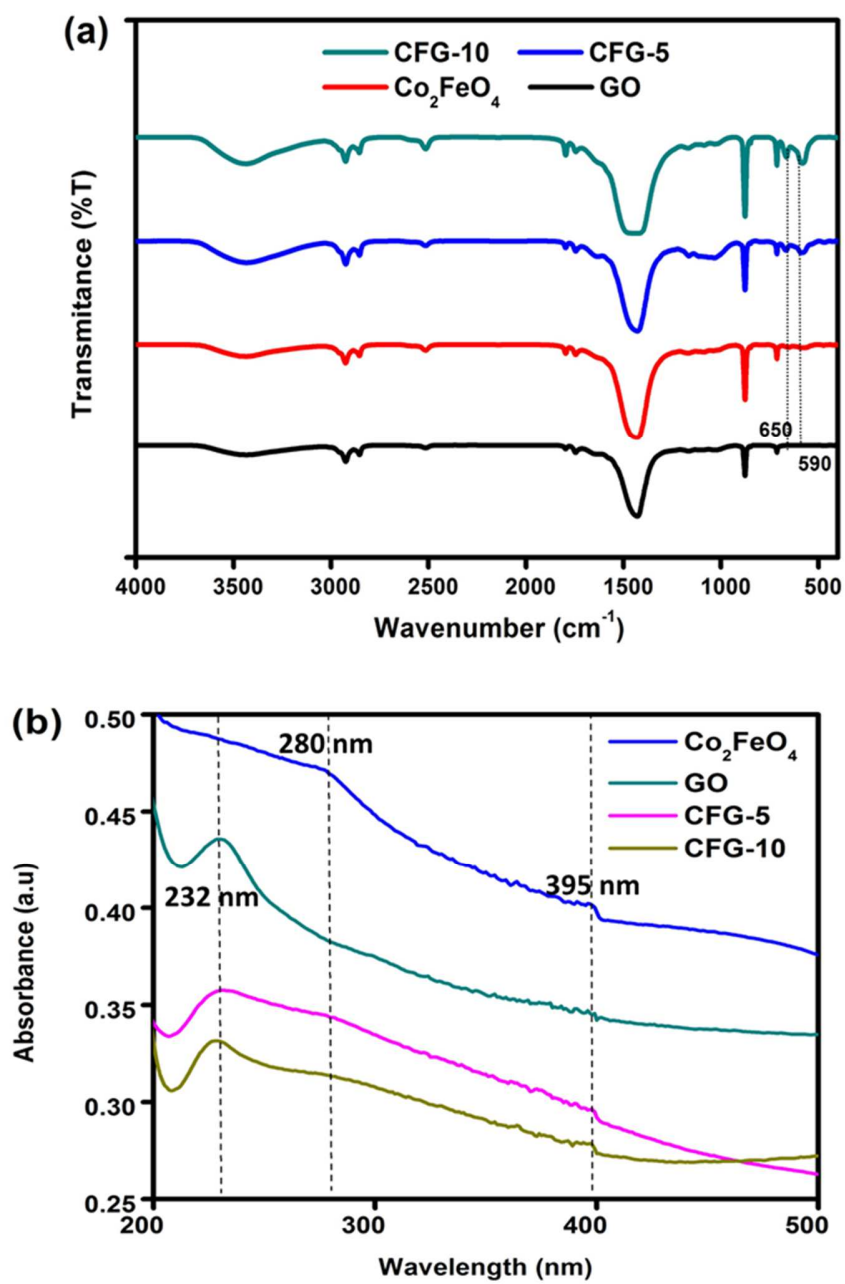


Fig. S3 (a) FTIR spectra of various nanostructures GO, Co_2FeO_4 , CFG-5 and CFG-10 (b) UV-vis analysis of as synthesized nanostructures

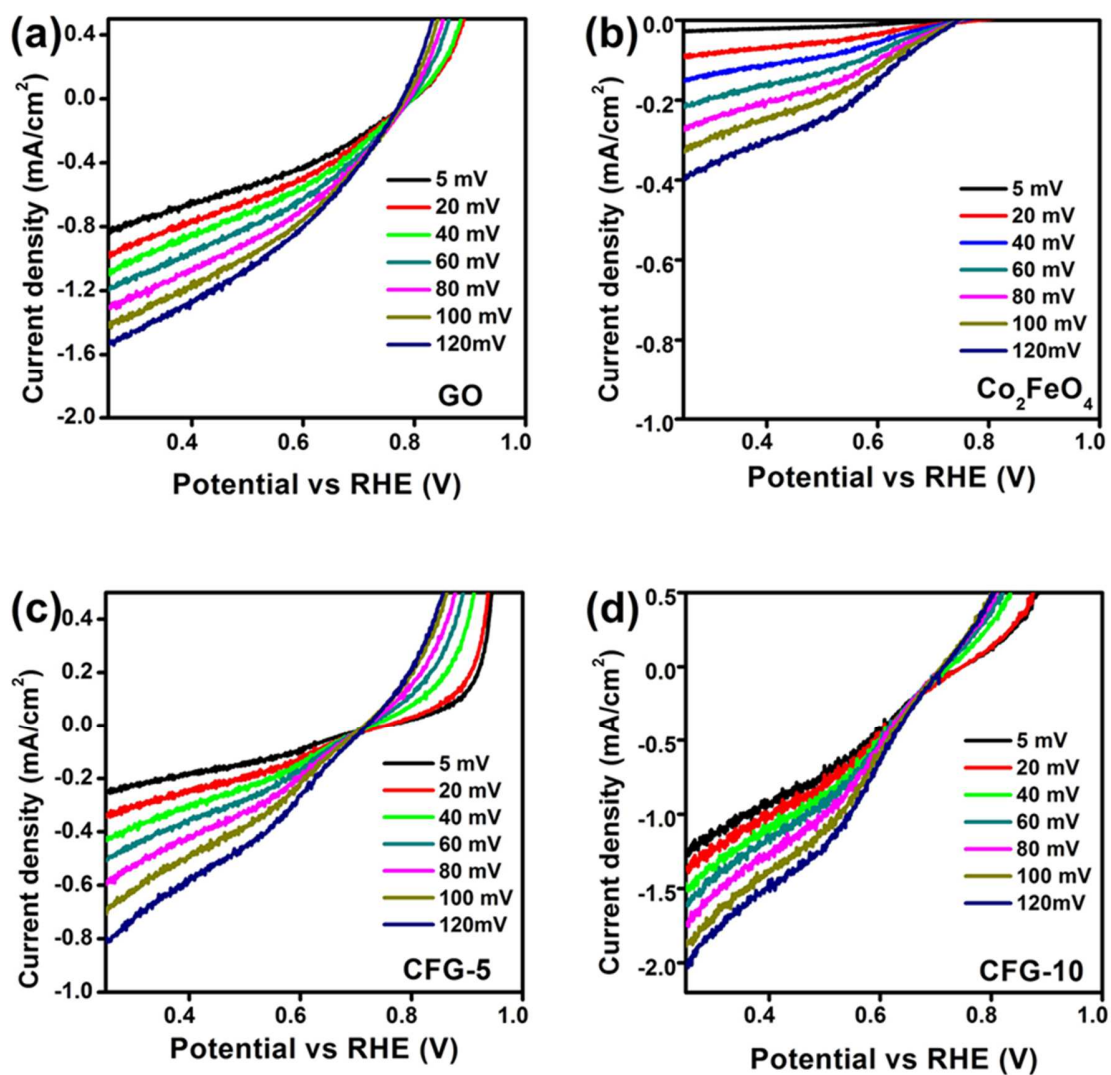


Fig. S4 Corresponding ORR activity of various nanostructures with varying scan rates at zero rotation speed

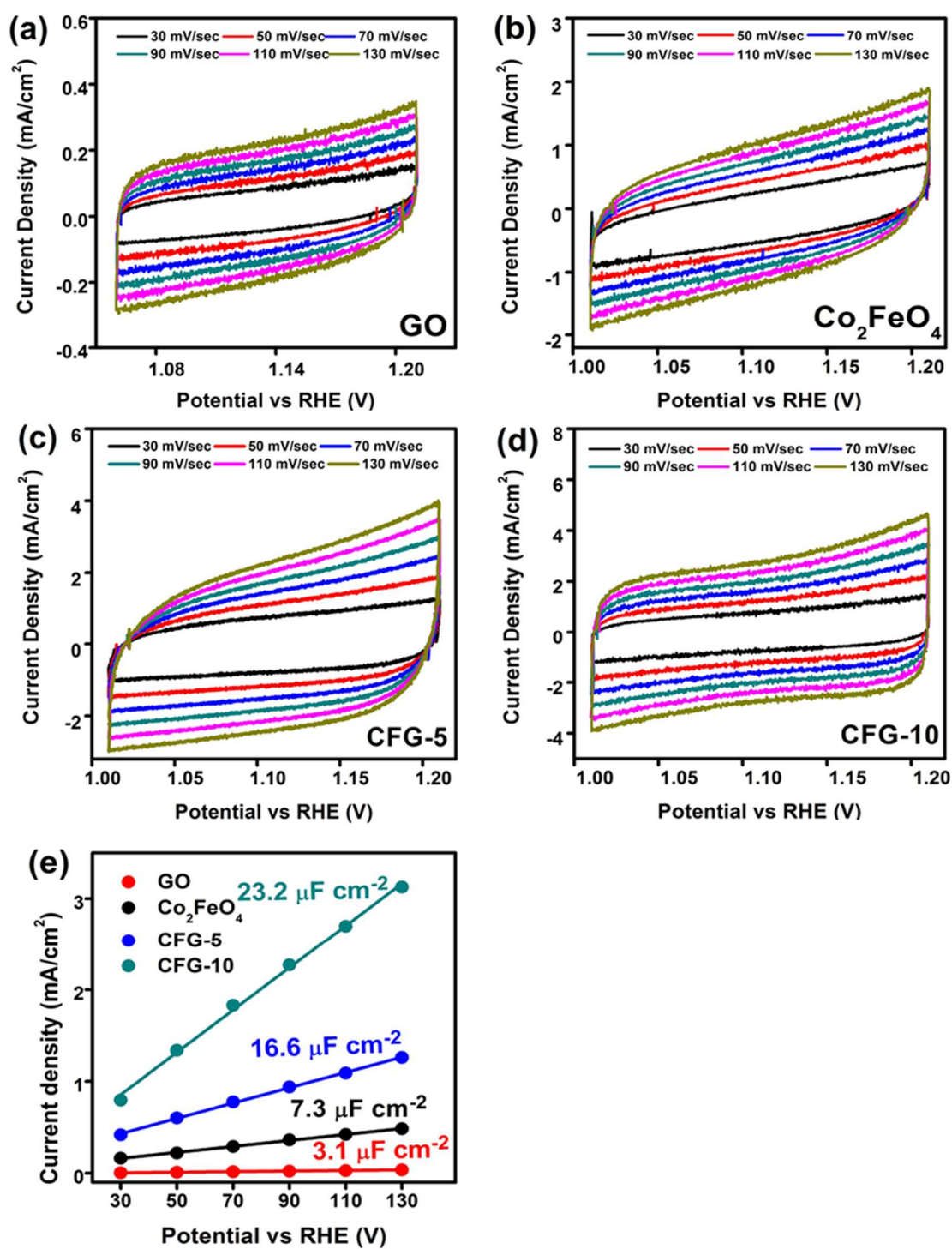


Fig. S5 CV curves of various nanostructures to determine corresponding double layer capacitance values (a) GO, (b) Co_2FeO_4 , (c) CFG-5 and (d) CFG-10

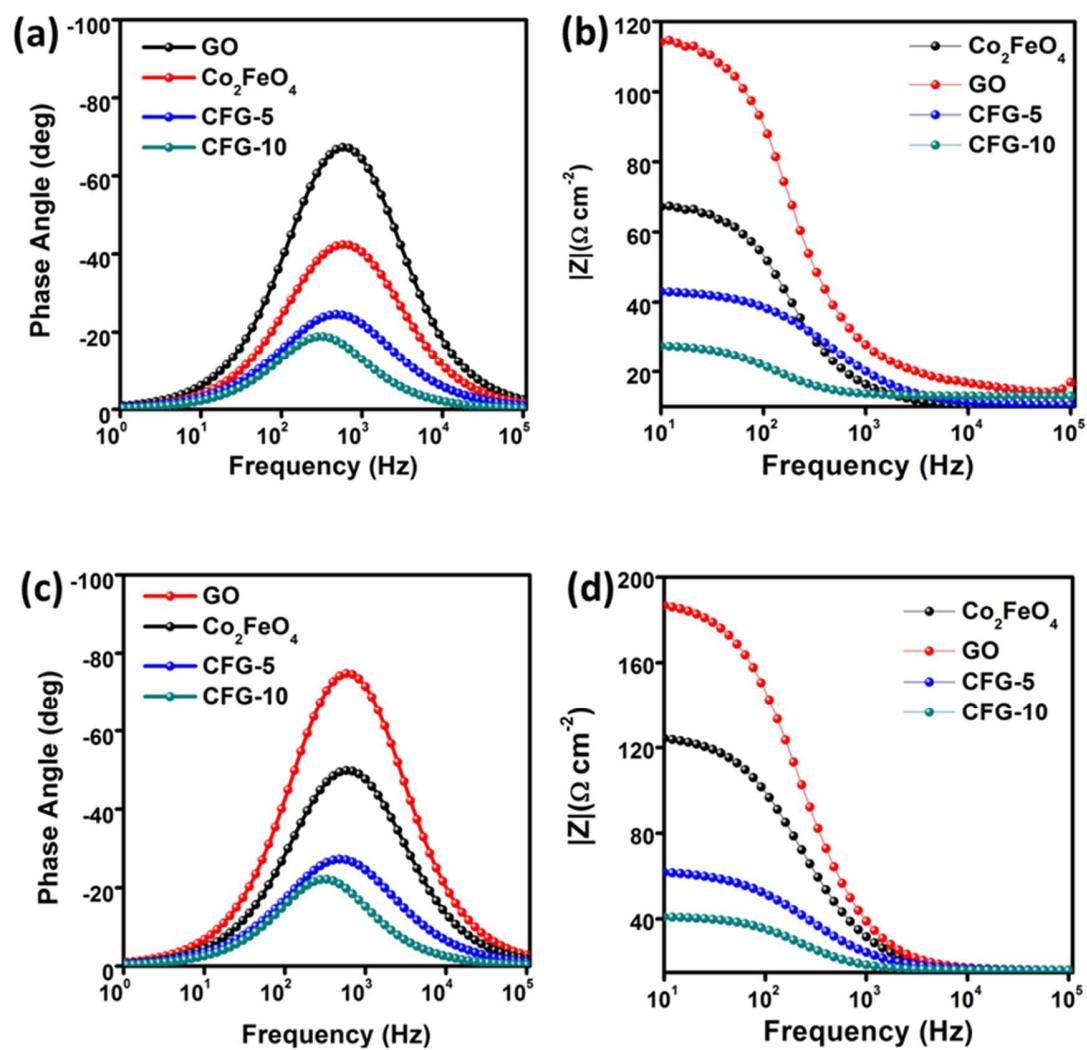


Fig. S6 Corresponding Bode plots (I and II) from EIS data (a and b) for OER analysis (c and d) for HER activity as well

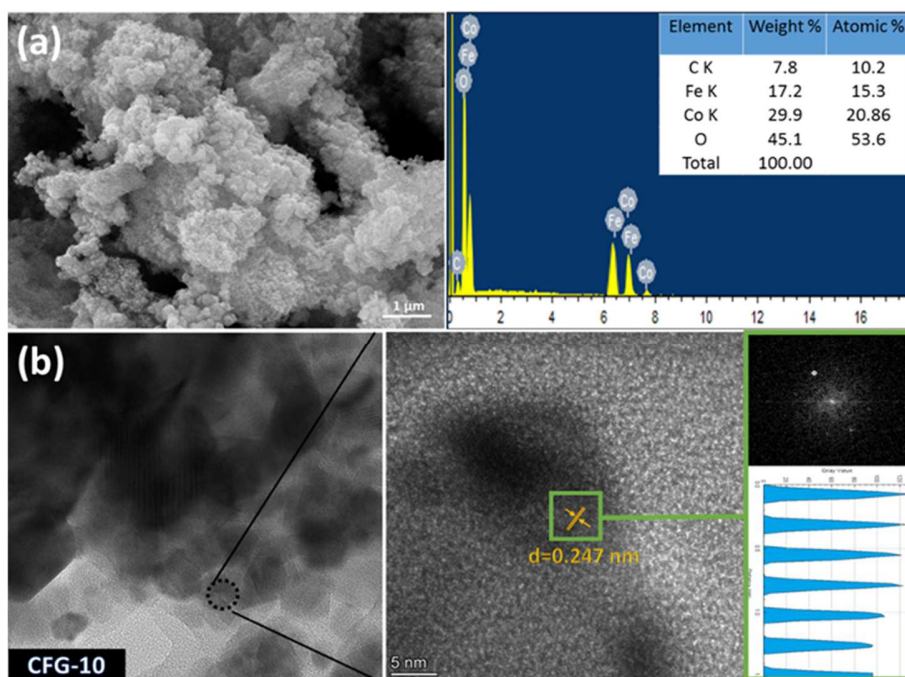


Fig. S7 After chronopotentiometry for OER the $\text{Co}_2\text{FeO}_4@\text{rGO}$ like CFG-10 composite was analyzed (a) SEM/EDS image (b) HRTEM image with relevant FFT image and lattice fringes information

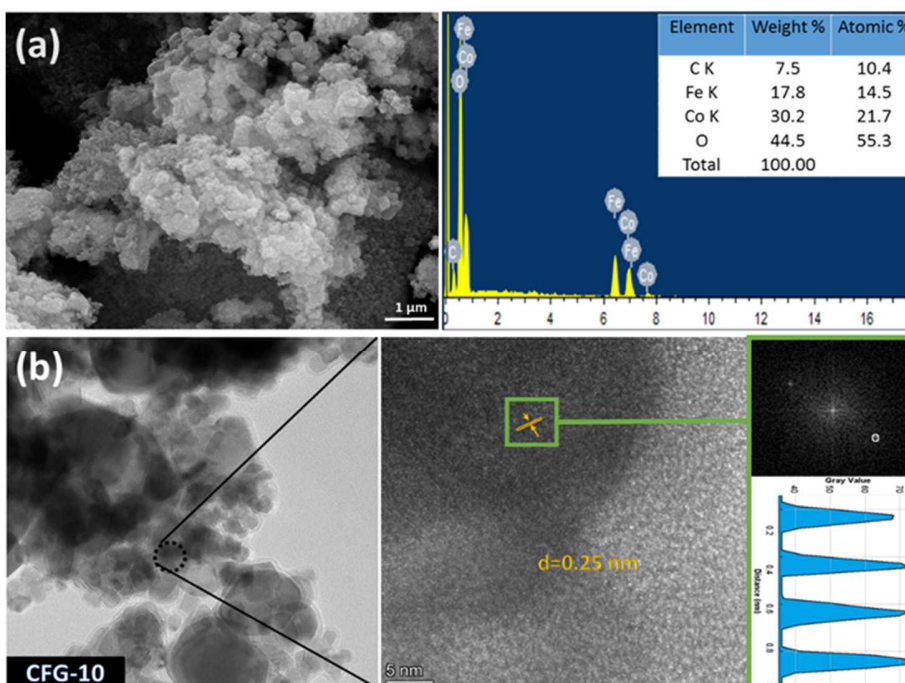


Fig. S8 After chronopotentiometry for HER the Co_2FeO_4 @rGO like CFG-10 composite was analyzed (a) SEM/EDS image (b) HRTEM image with relevant FFT image and lattice fringes information

Table S1. Comparative study of previous reported electrocatalysts for HER and current work.

| Electrocatalyst | Electrolyte | Current density | Overpotential | Tafel slope | Reference |
|--|--------------------|------------------------|----------------------|--------------------|------------------|
| CFG-10 | 1M KOH | 10 mA/cm ² | 320 mV | 48 mV/dec | This work |
| FeO _x -NBs | 0.5M KOH | 50 mA/cm ² | 450 mV | 85 mV/dec | [1] |
| NiFe ₂ O | 4.24M KOH | 10 mA/cm ² | 350 mV | 120 mV/dec | [2] |
| Ni _x Se/GO | 1M KOH | 30 mA/cm ² | 320 mV | 91 mV/dec | [3] |
| Fe-NiS/Ni(OH) ₂ | 1M KOH | 10 mA/cm ² | 350 mV | 95 mV/dec | [4] |
| Co ₃ O ₄ /Ppy/C | 1M KOH | 10 mA/cm ² | 340 mV | 87 mV/dec | [5] |
| Co@NGr | 1M KOH | 10 mA/cm ² | 340 mV | 146 mV/dec | [6] |
| Co _{1-x} Fe _x P ₃ | 1M KOH | 50 mA/cm ² | 330 mV | 65 mV/dec | [7] |

Table S2. Comparative study of previous reported electrocatalysts for OER and current work

| Electrocatalyst | Electrolyte | Current density | Overpotential | Tafel slope | Reference |
|---|--------------------|------------------------|----------------------|--------------------|------------------|
| CFG-10 | 1M KOH | 20 mA/cm ² | 240 mV | 51 mV/dec | This work |
| Co ₃ O ₄ @ZIF-67 | 1M KOH | 10 mA/cm ² | 330 mV | 73 mV/dec | [8] |
| Graphene-Co ₃ O ₄ | 1M KOH | 10 mA/cm ² | 313 mV | 49 mV/dec | [9] |
| CoP/rGO | 1M KOH | 10 mA/cm ² | 340 mV | 50 mV/dec | [10] |
| Co-TiO ₂ | 1M KOH | 50 mA/cm ² | 300 mV | 72 mV/dec | [11] |
| FeCoNiP | 1M KOH | 10 mA/cm ² | 350 mV | 60 mV/dec | [12] |
| Ni-Fe Oxide | 1M KOH | 60 mA/cm ² | 290 mV | 58 mV/dec | [13] |
| FeNiS2 NS/rGO | 0.1M KOH | 10 mA/cm ² | 270 mV | 40 mV/dec | [14] |

Table S3. Comparative study of previous reported electrocatalysts for ORR and current work

| Electrocatalyst | Electrolyte | Onset potential/ V vs RHE | Reference |
|-------------------------------------|--------------------|--------------------------------------|------------------|
| CFG-10 | 1M KOH | 0.60 | This work |
| MoS ₂ /N-graphene | 0.1M KOH | 0.95 | [15] |
| N,S-doped graphene | 0.1M KOH | 0.875 | [16] |
| N,P-doped graphene frameworks | 0.1M KOH | >0.845 | [17] |
| Co ₃ O ₄ /rGO | 0.1M KOH | 0.950 | [18] |
| Co/CoO _x | 0.1M KOH | 0.2 (vs AgCl) | [19] |
| rGO/Fe ₃ O ₄ | 0.1M KOH | 0.80 | [20] |
| Fe/Co-CNT | 0.1M KOH | 0.783 | [21] |
| Pt/AEI/rGO | 0.1M KOH | 1.05 | [22] |

Table S4. Comparative study of previous reported Tri-functional and Bi-functional electrocatalysts and current work.

| Electrocatalyst | Electrolyte | ORR | HER | | OER | | Ref. |
|---|-------------|------------------------------|-------------------------|----------------|-----------------------|----------------|------------------|
| | | Onset potential/ V vs RHE | Current density | Over-potential | Current density | Over-potential | |
| Co ₂ FeO ₄ @GO | 1 M KOH | 0.60 V | 10 mA/cm ² | 320 mV | 20 mA/cm ² | 240 mV | This work |
| D/G-CTS-n | 1 M KOH | 0.98 V | 10 mA/cm ² | 320 mV | 10 mA/cm ² | 420 mV | [23] |
| Co ₃ Mo ₃ N | 1 M KOH | 0.75 V | 10 mA/cm ² | 100 mV | 10 mA/cm ² | 290 | [24] |
| Co/CoO@CoN-C-800 | 0.1 M KOH | 0.78 V | 20 mA/cm ² | 436 mV | 10 mA/cm ² | 364 mV | [25] |
| Defect graphene | 0.1 M KOH | 0.76 V | 10 mA/cm ² | 320 mV | 10 mA/cm ² | 340 mV | [26] |
| Co _{0.85} Se@NC | 1 M KOH | 0.81 V | 10 mA/cm ² | 230 mV | 10 mA/cm ² | 320 mV | [27] |
| CoO _x NPs/BNG | 0.1 M KOH | 0.950 V | 10 mA/cm ² | 295 mV | - | - | [28] |
| YBaCo ₄ O _{7+d} | 0.1 M KOH | 0.82 V | 1.18 mA/cm ² | 400 mV | - | - | [29] |
| NiCoS _s | 1 M KOH | - | 20 mA/cm ² | 234 mV | 10 mA/cm ² | 168 mV | [30] |
| VoB-Co ₃ O ₄ /NF | 1 M KOH | - | 50 mA/cm ² | 184 mV | 50 mA/cm ² | 315 mV | [31] |
| Co ₃ O ₄ (3)/Ism-TiO ₂ | 1 M KOH | 0.84 V | - | - | 10 mA/cm ² | 348 mV | [32] |
| Co-N/S/rGO | 0.1 M KOH | 0.85 V | - | - | 10 mA/cm ² | 320 mV | [33] |

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